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# Single-atom engineered sensors for volatile organic compounds

Sowjanya Vallem a,b, Malayil Gopalan Sibi c, K. Keerthi d, Anam Giridhar Babu e, Vishaka Goyal c, EA Lohith d, N.V.V. Jyothi d, K. Praveena f, Kasibhatta Sivakumar f, T.G. Satheesh Babu a,b, P.V. Suneesh a,b, Hari Bandi a,b, Daniel-Ioan Stroe b, Sada Venkateswarlu c, Aristides Bakandritsos a, Rajenahally V. Jagadeesh c,j,\*\*, Radek Zboril c,j,\*\*

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#### ABSTRACT

The efficient and precise detection of trace-level volatile organic compounds (VOCs) is critically important for environmental monitoring, industrial safety, and public health. In this context, single-atom (SA) materials have emerged as a new frontier in sensor technology, offering unparalleled atom and energy efficiency, along with maximal exposure to active sites. Compared to conventional nanoparticle and bulk sensors, SA-based platforms exhibit superior sensitivity, selectivity, and tunability. This review presents a comprehensive overview of the advances in single-atom engineering (SAE) for VOC detection. We systematically discuss the design principles, fabrication methods, and sensing mechanisms of various SA-based sensors, including chemiresistive gas sensors (CGS), metal oxide semiconductors (MOS), microelectromechanical systems (MEMS), field effect transistors (FETs), and electrochemical sensors. Special attention is given to the roles of heteroatom doping, vacancy engineering, and support interactions in modulating the sensing performance. This review also highlights how advanced spectroscopic tools provide insight into SA-analyte interactions and how computational approaches, particularly density functional theory (DFT) and emerging machine learning (ML) techniques, aid in the rational design of next-generation sensors. Finally, we outline the current challenges and propose future research directions aimed at achieving scalable synthesis, long-term stability, and real-world deployment of SA-based VOC sensors. This review aims to guide future innovations in SA sensor technologies, setting the stage for transformative advances in VOC detection.

<sup>&</sup>lt;sup>a</sup> Department of Chemistry, Amrita School of Physical Sciences Coimbatore, Amrita Vishwa Vidyapeetham, Coimbatore 641112, India

<sup>&</sup>lt;sup>b</sup> Amrita Biosensor Research Laboratory, Amrita School of Engineering Coimbatore, Amrita Vishwa Vidyapeetham, India

<sup>&</sup>lt;sup>c</sup> Nanotechnology Centre, Centre for Energy and Environmental Technologies, VSB-Technical University of Ostrava, 17. listopadu 2172/15, Poruba, Ostrava 708 00, Czech Republic

<sup>&</sup>lt;sup>d</sup> Department of Chemistry, Sri Venkateswara University, Tirupati, Andhra Pradesh 517502, India

e Department of Basic Sciences, SR University, Warangal, Telangana 506371, India

f Department of Chemistry, Sri Venkateswara Arts College (TTD's), Tirupati, Andhra Pradesh 517502, India

g Department of Electronics and Information Convergence Engineering, Institute for Wearable Convergence Electronics, Kyung Hee University, 1732 Deogyeong-daero, Giheung-gu, Yongin-si, Gyeonggi-do 17104, Republic of Korea

<sup>&</sup>lt;sup>h</sup> Department of Energy, Aalborg University, Aalborg 9220 Denmark

i Regional Centre of Advanced Technologies and Materials, Czech Advanced Technology and Research Institute, Palacky University, Olomouc, Slechtitelu 27, Olomouc 77900. Czech Republic

<sup>&</sup>lt;sup>1</sup> Leibniz-Institut fur Katalyse e.V., Albert-Einstein-Str. 29a, Rostock D-18059, Germany

<sup>\*</sup> Corresponding author.

<sup>\*\*</sup> Correspondence to: Nanotechnology Centre, Centre for Energy and Environmental Technologies, VSB-Technical University of Ostrava, 17. listopadu 2172/15, Ostrava 708 00, Czech Republic

*E-mail addresses*: venkateswarlu.sada@vsb.cz (S. Venkateswarlu), a.bakandritsos@upol.cz (A. Bakandritsos), jagadeesh.rajenahally@catalysis.de (R.V. Jagadeesh), radek.zboril@upol.cz (R. Zboril).

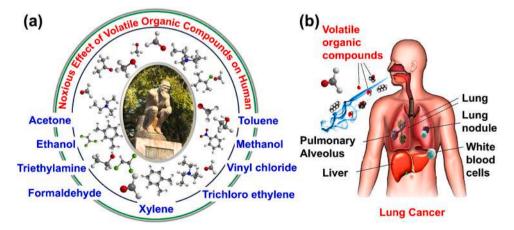
#### 1. Introduction

The rapid urbanization has led to a significant rise in the use of personal care products, cleaning chemicals, paints, solvents, air fresheners and others. The extensive utilization of these daily life products has resulted in the release of carbon-based volatile organic compounds (VOCs) into the atmosphere (both indoor and outdoor), which create substantial health and environmental risks [1-4]. According to the World Health Organization (WHO), VOCs such as formaldehyde, methyl (alkyl)amines, alcohols, benzene derivatives, light hydrocarbons and halogen compounds have adverse effects on human health, including an increased risk of leukemia, lung damage, dyspnea, throat infection, and other issues [5-7] (Scheme 1). The WHO and the European Union (EU) define VOCs as organic compound with a boiling point below 250 °C at 1 atm [8-10]. Given the concerns related to human health, environmental monitoring, and healthcare diagnostics, the development of eco-friendly, low-cost, and easily accessible sensors for the detection and estimation of VOCs is highly essential [11]. However, conventional detection methods such as gas chromatography and protein-based biosensors are often expensive, difficult to operate, and unsuitable for real-time monitoring of the diverse analytes [12–15].

In this context, gas sensors, including chemiresistive sensors, metal oxide semiconductor gas sensors, electrochemical sensors, and microelectromechanical system sensors, are among the most advanced devices for air quality monitoring [13-15]. Generally, most of the sensors are fabricated using n-type metal oxide nanoparticles (NPs) such as ZnO, In<sub>2</sub>O<sub>3</sub>, WO<sub>3</sub>, Fe<sub>2</sub>O<sub>3</sub>, and SnO<sub>2</sub> as well as p-type metal oxide nanoparticles like CuO, Cr<sub>2</sub>O<sub>3</sub>, and Co<sub>3</sub>O<sub>4</sub> [16-20]. However, pristine metal oxide nanoparticles based sensors suffer from low response toward the analytes, poor stability, and limited selectivity [21]. To address these limitations, researchers have developed heterostructures heterojunctions, such as SnO<sub>2</sub>/Fe<sub>2</sub>O<sub>3</sub>@rGO, SnO<sub>2</sub>/NiO, SnO<sub>2</sub>/Au@In<sub>2</sub>O<sub>3</sub>, and SnO<sub>2</sub>/ZnO. Similarly, sensors incorporating metal nanoparticles embedded in metal oxides (ZnFe<sub>2</sub>O<sub>4</sub>), or doped with noble metal nanoparticles (Pt, Pd, Au, Ru) have shown enhanced sensing capabilities and higher selectivity compared to non-noble metal catalysts. This improvement is primarily attributed to the high catalytic activity and electronic sensitization effects governed by the Schottky barrier mechanism [22-27]. However, several obstacles hinder the widespeared use of noble metal nanoparticle in gas sensor technologies. One key limitation is that only the outer surface atoms of the nanoparticles participate in the sensing reactions, which restricts atom utilization and reduces the overall response rate [28]. Additionally, issues such as limited availability and higher price as well as difficulties in controlling the active sites of noble metal nanoparticles, and particle-agglomeration remain significant challenges. The detection and removal of hazardous

pollutants using advanced nanomaterials has gained significant attention due to increasing environmental and health concerns. In addition to widely studied noble metal-based nanoparticles, the group led by Md. Rubel et al. explored earth-abundant mesoporous silica-based nanoparticles (NPs) functionalized with chelating organic ligands as effective sensor platforms for the selective detection and removal of toxic contaminants [29–42]. However, silica NPs often encapsulate or coordinate metal ions within organic ligands, which reduces the fraction of accessible metal active sites. Moreover, the insulating nature of silica and the static characteristics of organic ligands limit electronic modulation. Organic ligands may also introduce background signals or steric hindrance and typically rely on weak, non-specific interactions (e.g., chelation), thereby compromising the selectivity and sensitivity of the sensing process [43]. In our group, S. Venkateswarlu and colleagues have developed a variety of metallic nanoparticle-based catalysts, including Au/C carbon dots, Co<sub>3</sub>O<sub>4</sub>, ZrO<sub>2</sub>, graphene-based composites, metal-organic frameworks (MOFs), covalent organic frameworks (COFs), and carbon dot-based materials. These nanostructures have been employed in electrochemical and photoluminescence sensors for the detection of a broad spectrum of organic molecules such as dopamine, glucose, regorafenib, uric acid, and folic acid, as well as toxic heavy metal ions including Hg<sup>2+</sup> and Pb<sup>2+</sup> [44–49]. In our previous work, we also reported the detection of volatile organic compounds (VOCs), including ethylenediamine, diisopropylamine, and dioxane, using environmentally benign carbon onion-based catalysts [50]. Furthermore, we successfully utilized carbon dots as photoluminescent (PL) thermal sensors [51]. More recently, our research has shifted toward the development of next-generation single-atom-based sensors. In particular, we have demonstrated that copper single atoms exhibit excellent electrocatalytic activity for the sensitive and selective detection of dopamine [52]. These findings underscore the promising potential of single-atom catalysts (SACs) in the advancement of high-performance chemical sensing technologies.

Developing effective heterojunctions through heterogeneous systems can significantly enhance the catalytic activity and selectivity towards the VOCs, enabling ultra-low-level detection, selectivity, long-term stability, and green fuel production [53–56]. Owing to their maximum atom efficiency, low energy consumption, unique metal support interface, tunable coordination environments, and well-defined active sites, single atoms (SAs) facilitate effective adsorption and trigger the catalytic conversion of gas molecules, leading to high sensitivity in sensor technologies [57–59]. From the last decade, SAs have opened new avenues in chemical engineering, energy, catalysis, biotechnology, environmental science, and sensor technologies [60–68]. With respect to the applications in sensor-technology, X. Chen et al. investigated the electronic behavior of Sc and Ti SAs decorated on a graphdiyne catalyst for



Scheme 1. (a) Schematic representation of various volatile organic compounds (VOCs). (b) Noxious effects on human health such as lung cancer. Reprinted with permission from Ref. [7].

formaldehyde (HCHO) detection in 2017 and summarized the timeline of SAs based on various sensors, including SAES for VOC detection (Fig. 1a, b) [69]. Since then, SAE has been integrated with 1-dimensional (1D), 2-dimensional (2D), and 3-dimensional (3D) metal oxides, selenides, sulfides, metal-organic frameworks (MOFs), and MXenes to generate heterojunctions with abundant active sites such as PtSA@SnO<sub>2</sub>, Pt-Vo-W, RhSA@SnO<sub>2</sub>, AuSAs@SnO<sub>2</sub>, AuSA@In<sub>2</sub>O<sub>3</sub>, CoSA-CeO<sub>2</sub>@SnO<sub>2</sub>, CuSAs@WO<sub>2.7</sub>, RuSA@SnS<sub>2</sub>, PtSAs@ZnO, PtSA@MXene, and PdSAs@cMOF for the ultra-low detection of VOCs by CGS, MOS, FET, MEMS, and electrochemical techniques [70–74] (Scheme 3c,d).

Based on the increased applications of single atoms, many comprehensive reviews have been reported on SAs-based energy conversion, storage, and environmental applications [75–77]. In addition, few reviews are also focused on single-atom sensor technology related to purification of contaminated gases, biomedical, environmental monitoring, and food safety assays [57,58,68,78,79]. However, to the best of our knowledge, until now no comprehensive reviews have been published on the specific topic of single atom engineered sensors (SAES) for VOC detection (Scheme 2) [78,79]. Hence, we turned our interest to writing an in-depth review on the development and applications of single atom engineered sensors (SAES) for the detection of volatile organic compounds.

This review discusses the impact of coordination environments, heteroatom and defect engineering, as well as other physical parameters in SAES. We also highlight the integration of SAES with various low-dimensional substrates, ranging from 0D to 3D systems. To elucidate the synergetic interactions between SAs and target VOCs, advanced spectroscopic techniques play a key role. In this context, we summarize the characterization of SAES using in situ/operando diffuse reflectance infrared Fourier transform spectroscopy (DRIFTS), Raman spectroscopy, X-ray absorption near edge structures (XANES), and Extended X-ray absorption fine structure (EXAFS). Furthermore, we discuss the application of density functional theory (DFT) for mechanistic insights (Scheme 1d) and the use of advanced ML strategies to accelerate the development of SAES technology for VOC detection. Finally, we summarize current limitations and outline future opportunities for these



**Scheme 2.** Graphical representation of the overall concept of the SAs engineered VOCs sensor, operando techniques, DFT studies, and machine learning models.

multidimensional substrates supported by SAES to monitor hazardous VOCs and other environmental pollutants.

### 1.1. Scope of this review and SAES

Despite rapid advancements in single-atom catalysts (SACs) for energy conversion and environmental catalysis, their emerging application in chemical sensing, particularly for the detection of volatile organic compounds (VOCs), remains significantly underexplored and fragmented across isolated studies. VOCs are highly toxic, volatile, and environmentally persistent, posing serious health and ecological risks. Thus, the development of sensitive, selective, and stable detection platforms is of critical importance. Conventional VOC sensors, based on metal oxides, silica nanoparticles, or carbon-based composites, often

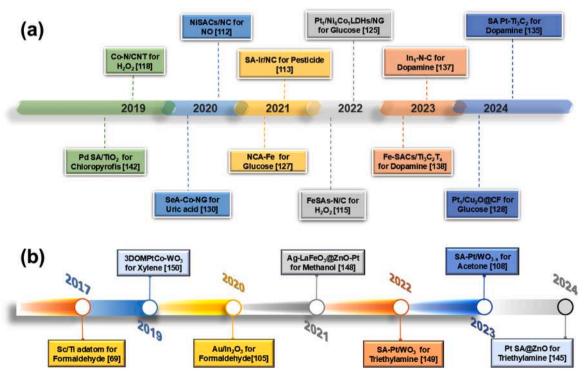


Fig. 1. (a) The timeline of single atom based various sensors. (b) Timeline of single-atom engineered VOCs sensors.

suffer from inherent drawbacks such as high operating temperatures, limited selectivity, and poor long-term stability under real-world conditions (Table 1) [80-87]. In contrast, SACs offer unique advantages, including maximum atomic efficiency, highly tunable coordination environments, and superior selectivity arising from their atomically dispersed active centers and strong metal support interactions. These characteristics position SACs as ideal candidates for next-generation chemical sensors. Although several reviews have comprehensively addressed the catalytic and biomedical applications of SACs, there is currently no focused and critical review that systematically explores their role in VOC detection. Considering the escalating demand for high-performance VOC sensors in indoor air quality monitoring, environmental surveillance, industrial safety, and wearable electronics, this review aims to fill this critical gap. The overall theme of the review is illustrated in (Scheme 3), which presents selected examples highlighting the synthesis of single-atom catalysts (SACs) supported on various morphologies, such as 2D MXene and nitrogen-doped carbon. It also depicts the working models of chemiresistive and electrochemical sensors designed for VOC detection. By consolidating recent progress, identifying existing challenges, and highlighting future opportunities, this work provides a timely and essential resource for advancing SAC-based VOC sensing technologies.

### 2. Single atom-based sensors

Single atoms in sensor technology have led to significant advancements due to their distinctive characteristics and advantages, including high atomic efficiency and tunable electronic properties. These features enable the development of highly sensitive, selective, and rapid detection sensors. SACs and single atom nanozymes are particularly fascinating for their ability to replicate natural enzymes with enhanced catalytic activity and selectivity [88,89].

Interestingly, SAs play a crucial role in biosensors, electrochemical sensors, and photoelectrochemical sensors, enabling precise detection of biomolecules and environmental analytes, even at ultra-low concentrations. Specifically, single-atom nanozymes have emerged as attractive biosensors due to their efficient atom utilization, enhanced selectivity, and stability, making them suitable for a wide range of biosensing applications. These nanozymes can mimic natural enzymes, which makes them valuable for detecting biological molecules, ions, organisms, and viruses [90]. Similarly, SAES demonstrates improved biomarker detection performance, broadening their applicability across various fields [57,91]. SAs are increasingly recognized as critical components of photoelectrochemical (PEC) sensors, enhancing sensitivity and selectivity. Researchers have developed novel biosensing technologies that utilize SACs to monitor biological substances in complex environments. These sensors, which integrate the unique properties of SAs with advanced PEC systems, deliver unprecedented performance across a wide range of applications [88,92].

#### 2.1. Heteroatom engineering of SAs

Heteroatom engineering of SAs involves the strategic insertion of heteroatoms (atoms other than the principal element in the system) into the structure of SACs [93]. This approach tunes the electrical, structural, and chemical properties of SACs, making them better suited for a variety of applications, including catalysis, energy devices, and sensors such as CGS, MOS, FFT, MEMS, and PEC.

Heteroatoms such as boron (B), nitrogen (N), phosphorous (P), oxygen (O), sulfur (S), and fluorine (F) in the support material or near the single atom influence the local coordination and electronic environment. They modify the charge distribution and d-band center of the metal atom, facilitate more active sites, and improve its catalytic and sensor activity [94]. The following recent developments highlight the importance of heteroatoms in SAE (Fig. 1).

A nitrogen-rich, high-surface conductive metal-organic framework (cMOF) has been employed as an effective single-atom support via a unique electrochemical deposition method (Fig. 2a). The generated Pd SAs are stabilized in a PdN<sub>4</sub> coordination by interplanar sites in the cMOF while preserving the porosity of the catalyst. This PdN<sub>4</sub> moiety, with its preserved porosity, enables high selectivity and sensitivity to the target gas analyte [95]. Another interesting approach involves the design of dual heteroatoms that have SACs, a guest B heteroatom is doped alongside the nitrogen moiety via a one-pot pyrolysis process (Fig. 2b). This method effectively tunes charge transfer and modulates the electronic structure of Fe atoms by generating various FeBNC coordination models, which lower the energy barrier and enhance the peroxide-like activity of single atom nanozymes [96].

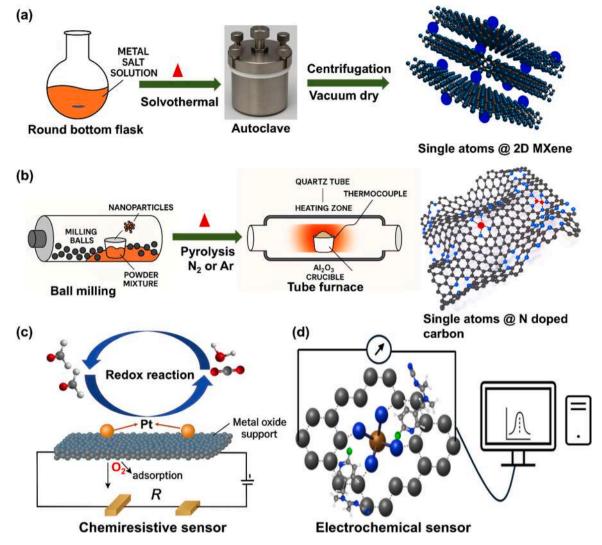
Similarly, a chemical etching coordination approach has been developed to enable the direct coordination of Fe-SAs with N and S heteroatoms (Fig. 2c). The coordination of the low electronegative S atom can tune the 3D orbital of Fe-SAs electronic structure and activate catalytic sites for more effective reaction [97]. Meanwhile, the two-dimensional graphitic carbon nitride (gCN) creates a stable coordination network with N moiety, which prevents the agglomeration of Pt SAs (4.3 wt%) even at elevated temperatures and provides more active sites, as shown in the STEM image (Fig. 2d) [72]. Beyond mono-and di-heteroatom strategies, H. Zhou et al. expanded single-atom engineering to include tri-heteroatom coordination. They developed an Mn-based single-atom porous carbon frame catalyst coordinated with N, S, and P (Fig. 2e). The incorporation of multiple heteroatoms introduces additional lone pairs of electrons, which stabilize Mn SAs. Advantageously, the high selectivity is attributed to the presence of multi-hierarchical pores in the as prepared Mn SACs [98].

In SAE, the use of environmentally benign heteroatoms, such as boron (B), in combination with biopolymers as nitrogen (N) atom

 Table 1

 Comparison of SACs, metal oxide nanoparticles, and organic ligand-chelated silica nanoparticles in sensor applications.

S. No	Parameter	Metal Oxide Nanoparticles (MONPs)	Organic Ligand- Chelated Silica NPs	Single-Atom Catalysts (SACs)	SAC Advantage	Ref
1	Metal atom utilization	Low-to-moderate (bulk or surface-bound)	Partial (due to buried/ chelating ligands)	100 % (all atoms are active sites)	Maximum atom efficiency	[80]
2	Conductivity	Generally low (e.g., TiO <sub>2</sub> , ZnO), may need conductive additives	Insulating silica core	High (with conductive supports: MXene, graphene)	Faster electron transfer	[81]
3	Chemical stability	High thermal and chemical stability	Moderate (ligands degrade in pH/UV)	Excellent (strong metal–support anchoring)	Tuned coordination gives both stability and selectivity	[82]
4	Selectivity	Moderate, depends on surface defect chemistry	Ligand-specific, often not robust	Tunable via atom type and coordination (e.g., Co–N)	Designer active sites enhance target recognition	[83, 84]
5	Detection limit	Moderate-to-low (affected by agglomeration and surface activity)	Moderate (ligand–analyte dependent)	Ultra-low (ppb to ppt)	Trace-level sensitivity due to unique adsorption	[85]
6	Harsh environment compatibility	Stable, but may suffer surface passivation	Weak (ligands degrade easily)	Stable in acidic, basic, ionic, oxidative media	Strong anchoring and inert supports resist deactivation	[86]
7	Multifunctionality	Redox-active, photocatalysis possible	Limited to ligand type (e. g., fluorescence)	Sensing, catalysis, and remediation	Versatile integration with other functions	[87]



**Scheme 3.** (a, b) Schematic illustrations of the synthesis of SACs via different methods and morphologies, including solvothermal and pyrolysis techniques. (c, d) Schematic representations of chemiresistive and electrochemical sensor models for VOC detection.

precursors, is an intriguing approach for developing eco-friendly single-atom catalysts. To design a high surface area hierarchical porous single-atom network, C. Xu et al. utilized boric acid and chitosan as B and N heteroatom sources. They synthesized B and N coordinated, high-loading Co SAs (4.2 wt%) within a hierarchical porous carbon framework via a one-pot pyrolysis strategy (Fig. 2f). Interestingly, at high temperatures, the formation of boron oxide prevents Co agglomeration, allowing for the uniform dispersion of isolated SAs with Co-N-B-C coordination. This also generates effective active sites for efficient electrochemical reactions [99].

The above SAE coordination strategies illustrate the significance of heteroatoms in preventing aggregation and leaching. Heteroatoms provide anchoring sites for SAs (such as N and S dopants in carbon supports), thereby stabilizing them and increasing the resilience of SACs under operating conditions [100]. Specific heteroatoms can synergize with SAs, enhancing reactant adsorption and reducing activation energy. Additionally, they may introduce defects that serve as additional active sites. By strategically manipulating the type and position of heteroatoms, SACs can be tailored to preferentially interact with specific reactants, thereby improving selectivity in sensor technology [101].

# 2.2. Defect engineering in single atom-based sensors

Defect engineering of SAs involves the deliberate introduction or

manipulation of atomic-level flaws in materials to stabilize or modify the properties of individual atoms. This emerging field is essential for advancing applications in catalysis, energy storage, and sensor technology [102]. Defect engineering includes creating and altering vacancies, as well as managing defects and grain boundaries in host materials such as carbon substrates, MXene, and metal oxides. For SAs, defect sites act as anchor points, stabilizing individual atoms or clusters on the substrate. Due to their high surface energy, SAs tend to agglomerate into larger clusters. Defects help to prevent agglomeration by creating high-energy sites that enhance interactions with the SAs and thus between the target molecules and the active sensing sites of the sensor material [103]. Various synthesis techniques, including plasma treatment, chemical etching, thermal annealing, and ultraviolet (UV)

irradiation [104], can create different types of defects, such as vacancy, substitutional, topological, strain-induced, and surface defects, all of which effectively increase the sensing properties (Fig. 3). On the other hand, metal oxide semiconductors for gas sensing have their practical application often limited by poor sensitivity, low selectivity, and high operating temperature requirements. Single atom engineering (SAE) presents a promising approach to overcome these challenges by enhancing the stability, selectivity, and sensitivity of metal oxide semiconductor gas sensors. By introducing SAs at defects or vacant sites in extremely low concentrations, activation energy and operating temperatures can be significantly reduced. This approach also provides

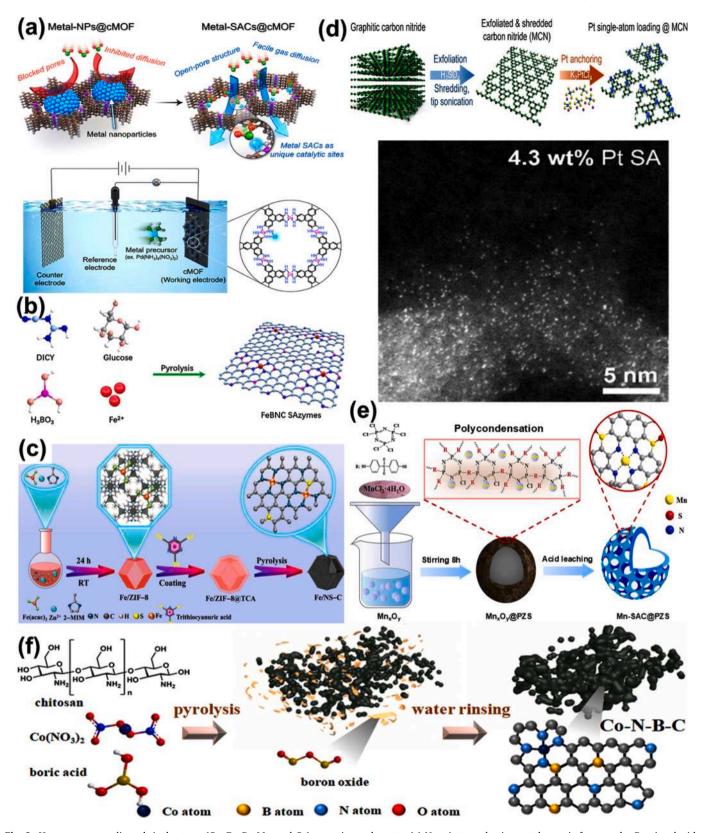


Fig. 2. Heteroatom- coordinated single atoms (Cu, Fe, Pt, Mn, and Co) on various substrates. (a) N moiety conductive metal-organic frameworks. Reprinted with permission from Ref. [95]. (b) B, N moiety carbon frame. Reprinted with permission from Ref. [96]. (c) N, S moiety carbon frame. Reprinted with permission from Ref. [97]. (d) N moiety graphitic carbon structure and STEM image of Pt SAs. Reprinted with permission from Ref. [72]. (e) N, S moiety porous carbon frame. Reprinted with permission from Ref. [98]. (f) B, N moiety has porous carbon frame Reprinted with permission from Ref. [99].

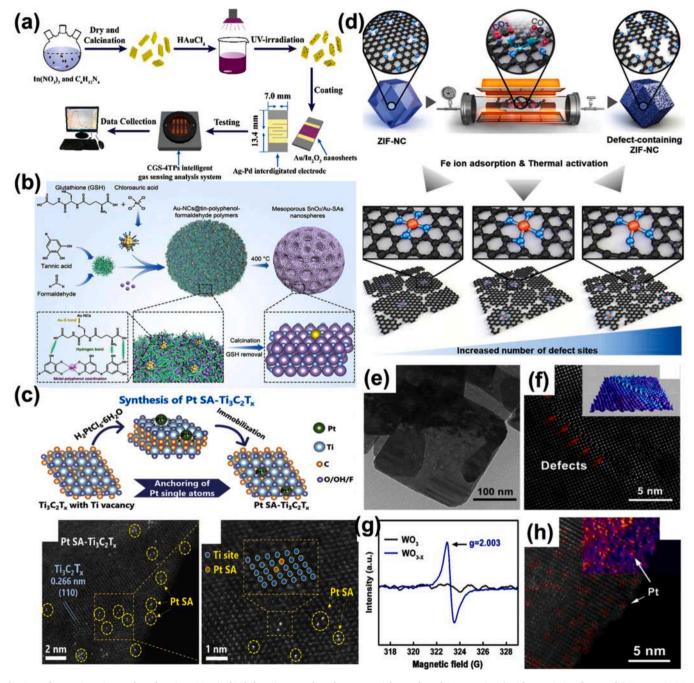


Fig. 3. Defect engineering anchored various SAs. (a, b) defect sites metal oxide nanoparticles anchored SAs Reprinted with permission from Ref. [105, 106]. (c) Pt SAs onto Ti vacancy positions of  $\text{Ti}_3\text{C}_2\text{T}_x$  (MXene) substrate and HAADF-STEM images. Reprinted with permission from Ref. [70]. (d) Defect site carbon substrate anchored Fe SAs. Reprinted with permission from Ref. [107]. (e-h) TEM and HAADF STEM images and EPR spectra of WO<sub>3</sub> nanoplate with oxygen vacancy defects can effectively accommodate Pt SAs. Reprinted with permission from Ref. [108].

abundant active sites with high atom efficiency, enabling ultra-low-level detection of target analytes. For instance, a novel UV light reduction method has been employed to dope  $\rm In_2O_3$  with a trace amount of Au SAs, significantly enhancing its sensing performance (Fig. 3a) [105]. In the Au SAs@In\_2O\_3 MOS sensor, Au SAs extract electrons from  $\rm In_2O_3$ , triggering an electron transfer phenomenon that enhances the performance of the MOS gas sensor. Similarly, introducing Au SAs into the interstitial sites of porous SnO\_2 with lattice deffects prevents their agglomeration while forming Au-O-Sn bonds, which provide additional active sites [106].

The Au-embedded porous SnO<sub>2</sub> improves sensing performance through both chemical and electric sensitization effects (Fig. 3b). These

defect engineering strategies effectively facilitate the fabrication of SAs metal oxide semiconductor gas sensors, enabling highly sensitive detection at lower working temperatures. B. Zong et al. successfully anchored Pt SAs onto Ti vacancy positions of  $Ti_3C_2T_x$  (MXene) substrate using a self-reduction immobilization method, which has been confirmed in HAADF-STEM images (Fig. 3c) [70]. The PtSAs@Ti\_3C\_2T\_x structure enhances chemical sensitization and gas adsorption, allowing for rapid detection of target gas molecules with high sensitivity. Another intriguing defect engineering approach involves tuning defect sites in a carbon substrate using a  $CO_2$  activation strategy (Fig. 3d) [107]. This method modulates the oxidation and spin states of Fe single atom centers, altering the electronic configuration of Fe-NC SAs and enhancing

their intrinsic catalytic activity. The role of defect engineering in single-atom growth is clearly emphasized by T. Yuan et al. In their study, they used WO $_3$  nanoplates as a substrate with and without oxygen vacancy defects. Fig. 3e and f display the TEM and AC-HAADF STEM images of WO $_3$  nanoplates with high oxygen vacancy defects, while the inset 3D image provides a clear visualization of the defect-engineered WO $_3$  nanoplate. Furthermore, the presence of oxygen vacancy defects in WO $_3$  nanoplate was confirmed by EPR spectra (g = 2.003) (Fig. 3g). Their findings revealed that while pure WO $_3$  tends to form nanoparticles, WO $_3$  nanoplates with oxygen vacancy defects can effectively accommodate Pt SAs, which are resistant against agglomeration (Fig. 3h). The presence of Pt SAs enhances the adsorption and lowers the reaction activation energy. The PtSAs@WO $_3$  sensor demonstrated exceptional selectivity (Spest/Specond = 2.8) and a detection limit as low as 0.1 ppm for VOC detection [108].

Precise defect management plays a crucial role in controlling a material's electronic structure, thereby enhancing the sensitivity of sensors to specific analytes. Materials with a high density of defects exhibit increased surface reactivity and faster charge transfer kinetics [109]. The advantages of defect-engineered SAs in sensing applications include high sensitivity and selectivity, lower detection limits, improved stability in challenging environments, and enhanced flexibility for detecting a wide range of analytes.

### 2.3. Other physical parameters

The physical characteristics of SAs in sensors play a crucial role in enhancing measurement sensitivity and precision. Recent advancements in quantum sensing and electrochemical applications have highlighted the unique properties of SAs, including their high activity at low mass content, strong interactions, and exceptional ability to detect minute forces and fields. Key aspects of SAES include force sensitivity, electric and magnetic field sensing, and material properties. The force sensitivity of SA sensors, particularly those using trapped ions, can reach subattonewton levels. These measurements involve super-resolution imaging, which enables precise three-dimensional displacement detection [110]. Quantum sensors composed of SAs can monitor electric and magnetic fields at the atomic scale, facilitating the study of spin textures and interactions in quantum materials [111]. Additionally, single-atom materials have demonstrated enhanced electrochemical sensing capabilities, including improved sensitivity, selectivity, and stability. The careful design of these materials is crucial for optimizing their sensing performance, making them suitable for a wide range of applications.

# 3. Design of single atom-based sensors for the detection of VOCs

Accurate detection of VOCs is essential for ecological safety, healthcare, and disaster management, such as locating victims through VOCs emitted during human decomposition. Traditional detection methods, though highly accurate, rely on large and bulky laboratory instruments, making them unsuitable for on-site or real time applications. To address these limitations, portable alternatives, such as electronic noses (e-noses), have been developed. These devices attempt to mimic the mammalian olfactory systems using multi-sensor arrays that integrate optical, acoustic, semiconductive polymer, and electrochemical sensing technologies. However, despite their portability, enoses often suffer from limited specificity, which hinders their particle deployment in complex environments. Several commonly used sensor technologies, MOS sensors, conducting polymer (CP) sensors, surface acoustic wave (SAW) sensors, and quartz crystal microbalance (QCM) sensors, operate through distinct mechanisms and offer unique advantages and limitations. MOS sensors, which function via a chemiresistive mechanism, are cost-effective and sensitive but typically exhibit poor selectivity and diminished performance in the presence of complex gas mixtures. CP sensors are capable of room-temperature operation and demonstrate high sensitivity, yet they are susceptible to environmental

instability, particularly under varying humidity levels. SAW sensors, while offering high sensitivity and compact design, face issues with temperature fluctuations and material degradation. QCM sensors detect changes in mass with high precision but require advancements in material stability to achieve commercial viability. In addition, photoionization detectors (PIDs) and electrochemical sensors are also key technologies for VOC detection. PIDs offer broad-spectrum detection capabilities but suffer from poor selectivity. EC sensors, which rely on redox reactions facilitated by catalytic materials such as various inorganic nanoparticles, are relatively selective but can face challenges such as baseline drift and cross-sensitivity to other gases.

To address the limitations of conventional sensors, SACs have emerged as a promising advancement (Table 2) [112–143]. SACs significantly enhance sensor performance due to their unique catalytic properties, which improve both sensitivity and selectivity. When anchored onto MOS surfaces, SAs facilitate efficient electron transfer, thereby optimizing sensor responsiveness and accuracy. This innovation highlights the potential of SACs to address the inherent challenges associated with traditional sensing materials.

This review explores the fundamental principles of SAC-based VOC sensors, including their sensing mechanisms, integration with various substrates, and applications in advanced sensing platforms such as chemiresistive, electrochemical, FETs, and MEMS. The integration of SACs into these platforms offers the way for the development of highly sensitive, selective, and portable VOC detection systems suitable for real-world applications.

#### 3.1. Working principle and significance of SAE VOCs sensor

Chemiresistive sensors represent a versatile and widely adopted class of devices for VOC detection, primarily leveraging electron transfer mechanisms during gas sensor interactions. Their popularity stems from advantages such as low power consumption, high sensitivity, and compatibility with advanced nanomaterials. The progression of chemiresistive sensor technology has been significantly driven by advances in material science and nanotechnology, resulting in improved selectivity, sensitivity, and operational stability.

Traditional chemiresistive sensors typically operate through the catalytic activity of sensing materials, wherein adsorbed gas molecules undergo oxidation or reduction reactions, leading to measurable changes in electrical signals. For example, metal oxide-based (MOX) chemiresistive sensors detect resistance changes upon gas adsorption and are classified as n-type or p-type, depending on the dominant charge carriers. Similarly, CP-based sensors, such as those using polyaniline or polypyrrole, exploit reversible electronic or structural changes in response to VOC exposure. Despite their effectiveness, these conventional systems often suffer from limitations, including high operating temperatures, cross-sensitivity to other gases, and long-term performance degradation. To address these challenges, hybrid sensor architectures have emerged as promising alternatives. These designs integrate MOX materials with porous frameworks such as carbon allotropes, boron nitride nanotubes, and MOFs, and 2D materials like transition metal carbides (MXenes) and chalcogenides. By combining the complementary properties of these components, such hybrid systems enhance gas adsorption capacity, promote efficient electron transfer, and improve operational robustness under varying humidity and temperature conditions.

Among these advancements, SAs embedded in MOX, porous carbon frameworks, or MOFs have significantly improved sensor selectivity and stability. These systems enable rapid and precise detection of specific VOCs. The enhancement arises from the unique characteristics of single-atom active sites, which provide exceptionally high surface-to-volume ratios, tunable electronic structures, and superior catalytic activity. The evolution of VOC sensors, propelled by advances in material science and nanotechnology, has enabled the development of efficient, selective, and reliable VOC detection mechanisms. The integration of SACs

**Table 2**Summary table of various SAC-based catalytic sensors for detecting multiple analytes.

S.No.	Category	Analyte	Catalyst	<b>Detection Limit</b>	Linear range	Ref
1.	Electrochemical sensor	NO	Ni SACs/N-C	$1.8 \times 10^3 \mu M$	~1.3 µM	[112]
2.	Electrochemical sensor	OPs	SA-Ir/NC	$0.17 \text{ ng ML}^{-1}$	0.5-500 ng ML <sup>-1</sup>	[113]
3.	Electrochemical sensor	$H_2O_2$	Fe-SASC/NW	$46.35 \times 10^{-9} \text{ m}$	$5.0 \times 10^{-10}$ m to 0.5 m	[114]
4.	Electrochemical sensor	$H_2O_2$	FeSAs-N/C	0.34 μΜ	764–9664 μM	[115]
5.	Electrochemical sensor	$H_2O_2$	Fe <sub>3</sub> C@C/Fe-N-C	0.26 μΜ	1–6000 μΜ	[116]
6.	Electrochemical sensor	$H_2O_2$	Co-N-C-800	0.13 μΜ	$0.31 \times 10^4  \mu\text{M}$	[117]
7.	Electrochemical sensor	$H_2O_2$	Co-N/CNT	0.0324 μΜ	$0.055\times10^4~\mu\text{M}$	[118]
8.	Electrochemical sensor	As(III)	Co SAC	0.02 ppb	0.1-3 ppb	[119]
9.	Electrochemical sensor	As(III)	Pt <sub>1</sub> /MoS <sub>2</sub>	0.05 ppb	0.5-8 ppb	[120]
10.	Electrochemical sensor	As(III)	FeN <sub>3</sub> P	0.01 ppb	0.5–8 ppb	[121]
11.	Electrochemical sensor	Pb(II)	Mn/g-C <sub>3</sub> N <sub>4</sub>	0.01 μΜ	0.1-0.8 μΜ	[122]
12.	Electrochemical sensor	HQ	Mo SAs/NPO-C	0.005 μΜ	0.02-200 μΜ	[123]
13.	Electrochemical sensor	Nitrobenzene	Nb-BNC	0.70 μΜ	2–100 μΜ	[124]
					100-600 μΜ	
14.	Electrochemical biosensor	Glucose	Pt <sub>1</sub> /Ni <sub>6</sub> Co <sub>1</sub> LDHs/NG	10 μΜ	10-2180 μM	[125]
15.	Electrochemical biosensor	Glucose	NCA-Co	0.1 μΜ	0.5–1000 μΜ	[126]
16.	Electrochemical biosensor	Glucose	NCA-Fe	0.5 μΜ	2-2000 μΜ	[127]
17.	Electrochemical biosensor	Glucose	Pt <sub>1</sub> /Cu <sub>2</sub> O@CF	1 μΜ	2-560 μΜ	[128]
18.	Electrochemical biosensor	L-dopa	PtHPCN-222	0.003 μΜ	0.1–1 μM	[129]
19.	Electrochemical biosensor	Uric Acid	A-Co-NG	0.0333 μΜ	0.4–1055 μΜ	[130]
20.	Electrochemical biosensor	Uric Acid	Ru <sub>3</sub> /NC	0.01 μΜ	0.05-1000 μM	[131]
21.	Electrochemical biosensor	Uric acid	Fe-N <sub>5</sub>	27 μΜ	0.01-480 μΜ	[132]
22.	Electrochemical biosensor	Uric acid	Co-N-C/rGA	3.59 μM	4–300 μΜ	[133]
23	Electrochemical biosensor	Uric Acid	Ru-Ala-C <sub>3</sub> N <sub>4</sub>	170 μΜ	0.5–2135 μΜ	[134]
24.	Electrochemical biosensor	Uric Acid	SA Pt-Ti <sub>3</sub> C <sub>2</sub>	0.0125 μΜ	3.5-5.8 μM	[135]
25.	Electrochemical biosensor	Dopamine	Ni-MoS <sub>2</sub>	$1  imes 10^{-6}~\mu M$	$1\times1061000~\mu\text{M}$	[136]
26.	Electrochemical biosensor	Dopamine	SA Pt-Ti <sub>3</sub> C <sub>2</sub>	0.0022 μΜ	4.3–5.7 μM	[135]
27.	Electrochemical biosensor	Dopamine	Fe-N <sub>5</sub>	$7  imes 10^{-6}~\mu M$	0.005-500 μM	[132]
28.	Electrochemical biosensor	Dopamine	In <sub>1</sub> -N-C	0.279 μΜ	0–500 μΜ	[137]
29.	Electrochemical biosensor	Dopamine	Co-N-C/rGA	0.32 μΜ	2-80 μΜ	[133]
30.	Electrochemical biosensor	Dopamine	Fe-SACs/Ti <sub>3</sub> C <sub>2</sub> T <sub>x</sub>	0.001 μΜ	0.01-200 μΜ	[138]
31.	Electrochemical biosensor	Dopamine	Mn-MoS <sub>2</sub> /PGS	$5  imes 10^{-5}~\mu M$	$5\times10550~\mu\text{M}$	[139]
32.	Electrochemical biosensor	Dopamine	Co-N-C SANs	0.04 μΜ	0.06-1200 μM	[117]
33.	Electrochemical biosensor	Dopamine	Ru-Ala-C <sub>3</sub> N <sub>4</sub>	20 nM	0.06-490 μM	[134]
34.	Electrochemical biosensor	Homo-vanillic acid	Fe-SACs/Ti <sub>3</sub> C <sub>2</sub> T <sub>x</sub>	0.01 μΜ	0.02-200 μΜ	[138]
35.	Electrochemical biosensor	Vanillylmandelic acid	FeSACs/Ti <sub>3</sub> C <sub>2</sub> T <sub>x</sub>	0.005 μΜ	0.01-200 μΜ	[138]
36.	Electrochemical biosensor	Ascorbic acid	Rh SAzymes	0.26 μΜ	10.0-53, 100 μM	[140]
37.	Electrochemical biosensor	Furazolidone	Pd <sub>1</sub> /NC	0.0333 μΜ	0.01–50,	[141]
					50–300 μΜ	
38.	Photoelectrochemical sensor	Chlorpyrifos	Pd SA/TiO <sub>2</sub>	0.01 ng/ML	$0.03$ ng/ML to $10 \mu$ g/ML	[142]
39.	Photoelectrochemical sensor	Prostate-specific antigen	SA Pt anchored Zn <sub>0.5</sub> Cd <sub>0.5</sub> S	0.22 pg/ML	1.0-10000 pg/ML	[143]

Abbreviations: OPs: Organophosphorus pesticides, HQ: hydroquinone

within MOX, porous materials, and 2D metal carbides as well as chalcogenides exemplifies the potential of hybrid approaches in revolutionizing sensor design. These strategies effectively address key challenges such as cross-sensitivity, high operating temperatures, and long-term stability [71–73]. As a result, VOC sensors are emerging as indispensable tools for environmental monitoring, healthcare diagnostics, and the development of next-generation portable technologies (Scheme 3c, d).

#### 3.2. Metal oxide semiconductor-based SAE sensors

### 3.2.1. 1D material-supported SAE sensors

One-dimensional (1D) materials, such as nanotubes and nanowires, offer exceptional platforms for supporting SAs due to their high surface area, excellent conductivity, and mechanical flexibility. Anchoring SAs onto 1D materials enables the creation of hybrid sensors with enhanced charge transport properties and increased active sites for VOC interactions. This synergy makes 1D-supported SA sensors ideal for applications requiring high sensitivity and flexibility, such as wearable electronics. By optimizing SA placement on 1D structures, these sensors achieve rapid response times and robust performance.

Initial studies primarily focused on SACs supported by MOXs. MOX materials offer a broad set of features that make them ideal for SAES. Their high surface area, intrinsic oxygen vacancies, and ability to stabilize SAs via strong metal-support interactions provide a stable platform for supporting SAs. Moreover, MOXs can exhibit excellent

electronic conductivity, thermal stability, and environmental durability, which are critical for long-term sensor performance. These materials also modulate the electronic structure of the anchored SAs, thereby finetuning their reactivity and selectivity toward specific analytes. The synergy between MOXs and SAs in sensor applications arises from their complementary roles. While SACs provide atomically dispersed active sites with high catalytic efficiency, MOXs serve as a dynamic interface for analyte adsorption, charge transfer, and reaction activation. For instance, the presence of surface oxygen vacancies in MOXs enhances their adsorption capability for gas molecules, facilitating electron transfer processes that are crucial for sensor response. Furthermore, the combination of SAs with MOXs enables the design of sensors with low detection limits, fast response times, and excellent resistance to interference, even under complex environmental conditions [68]. Various synthetic approaches have been employed to prepare 1D MOX supports, including the use of composites where MOXs are decorated or co-impregnated with SAs on nanotubes, polymer-derived nitrogen-doped carbon, and methods to fabricate metal oxide nanowires.

For instance, incorporating WO $_3$  into Pt/BN catalysts significantly enhances propane oxidation performance by introducing new active sites at the Pt-WO $_3$  interface [144]. These interfacial sites create a synergistic environment where propane ( $C_3H_8$ ) adsorption and C-H bond activation are facilitated. The reaction mechanism is governed by the interaction between  $C_3H_8$  adsorbed on Pt and surface hydroxyl groups on WO $_x$ , which accelerates the rate-determining step and C-H bond cleavage. The Pt-WO $_3$  interface modifies the electronic structure of the

Pt species and introduces novel active sites. This interaction alters the reducibility and surface acidity of the catalyst, which directly impacts  $C_3H_8$  adsorption and activation. Specifically, the  $WO_3$  sites adjacent to Pt provide hydroxyl groups that participate in the oxidation reaction, enhancing the cleavage of C-H bonds and facilitating the subsequent oxidation steps. The promotional effect of  $WO_3$  is evident from the significantly higher intrinsic activity of Pt-7 W/BN (7 wt% of W) compared to Pt/BN. The apparent activation energy  $(E_a)$  for the  $WO_3$ -containing catalyst is reduced, suggesting an alternate reaction pathway. The reaction orders of  $C_3H_8$  ( $\sim$ 1) and oxygen (negative) underscore the importance of oxygen coverage on the Pt surface, which is modulated by the  $WO_3$ . Reduced oxygen inhibition on the Pt-7 W/BN catalyst indicates that the Pt-WO $_3$  interface mitigates excessive oxygen adsorption, thereby enhancing  $C_3H_8$  activation. XPS analysis reveals that metallic Pt species alone are insufficient to account for the observed

activity, highlighting the dominant role of  $Pt^{\delta^-}$  species at the Pt-WO<sub>3</sub> interface [144]. The interplay between enhanced  $C_3H_8$  activation, increased reducibility, surface acidity, and stable active sites contributes to the overall promotional effects. These results underscore the potential of MOX-promoted catalysts for efficient light alkane oxidation.

A notable example reported by Shin et al. involved polymer-derived nitrogen-doped carbon, where Pt SAs were impregnated onto  $SnO_2$  nanowires using a melamine-based sacrificial carbon nitride (MCN) template [72]. Electrospinning was employed to form nanofibers, which were subsequently converted into a tubular structure. The post-calcination ensured uniform Pt dispersion within the  $SnO_2$  lattice. XPS analysis confirmed the presence of  $Pt^{4*}$ , validating the successful incorporation of Pt SAs.

The sensing performance of the designed SA-based sensor for formaldehyde (HCHO) was systematically evaluated and compared against

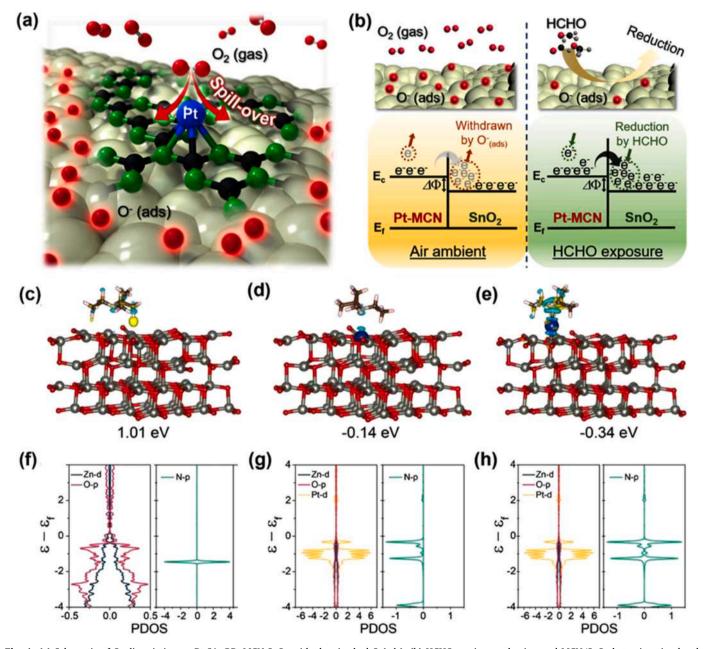


Fig. 4. (a) Schematic of  $O_2$  dissociation on Pt SAs@Pt-MCN-SnO<sub>2</sub> with chemisorbed O<sup>-</sup>(ads). (b) HCHO-sensing mechanism and MCN/SnO<sub>2</sub> heterojunction band diagram Reprinted with permission from Ref. [72]. Optimized geometries of TEA adsorption on (c) Pt@ZnO with an oxygen vacancy. (d) Pt@ZnO with Zn substitution by Pt, and (e) Pt SAs on ZnO near an oxygen vacancy. (f-h) Corresponding DFT-calculated band structures and DOS Reprinted with permission from Ref. [145].

Pt-doped SnO $_2$  nanofibers (Pt-SnO $_2$ ), MCN-SnO $_2$  without Pt, and pristine SnO $_2$ . The SA-based sensor exhibited 2.2-fold and 1.4-fold higher responses than pristine SnO $_2$  and MNC-SnO $_2$ , respectively, owing to enhanced catalytic activity and heterojunction formation. Among these, Pt-MCN-SnO $_2$  demonstrated the best performance, with a response (Rair/Rgas) of 33.9 at 5 ppm HCHO. This enhancement was attributed to synergistic effects, including the maximized atomic efficiency of Pt, heterojunctions between MCN and SnO $_2$ , and morphological optimization. Mechanistically, the Pt-N/C active site enhances O $_2$  adsorption

and dissociation (Fig. 4a), evidenced by an adsorption energy of  $-1.93~\rm eV$  and an elongated O–O bond compared to bulk Pt. This increases the concentration of chemisorbed oxygen species, facilitating HCHO adsorption and reaction. Electron transfer from Pt-MCN to SnO<sub>2</sub> reduces the work function (from 4.48 eV to 4.38 eV) and forms electron accumulation layers, amplifying resistance variations upon HCHO exposure (Fig. 4b). The increased baseline resistance of the sensor was supported by XPS data showing a higher O'/O<sup>2-</sup> ratio (71.4 %) in Pt-MCN-SnO<sub>2</sub> compared to pristine SnO<sub>2</sub> (46.7 %).

Experimental validation of the PtNP-MCN-SnO<sub>2</sub> material confirmed the superior performance of atomically dispersed Pt, attributed to enhanced active site exposure and suppressed agglomeration. Furthermore, Pt-MCN-SnO<sub>2</sub> exhibited exceptional stability, maintaining

atomically dispersed Pt after 170 h of operation at  $275^{\circ}$ C. The material showed only a 7.1 % decrease in HCHO response after two months of storage, compared to 30.0 % for Pt-SnO<sub>2</sub> and 54.1 % for pristine SnO<sub>2</sub>. This stability was ascribed to dual stabilization provided by MCN and SnO<sub>2</sub> nanograins, which prevented Pt migration and aggregation. These findings highlight the potential of N-doped carbon and MOX-supported SACs for high-performance, durable gas sensors, offering scalable synthesis and tunable selectivity for diverse VOCs applications.

Next, Pt SAs supported on ZnO nanorods were synthesized via a hydrothermal method for triethylamine (TEA) detection [145]. Notably, the Pt@ZnO sensor achieved an ultra-high response value of 4170 at 200 °C, with rapid response and recovery times of 34 and 76 s, respectively, and a low detection limit of 7.5 ppb. The enhanced sensitivity and specificity of SACs stem from their atomic dispersion, which maximizes active sites for gas interaction. In Pt@ZnO sensors, surface oxygen vacancies play a critical role in activating adsorbed oxygen species such as O and O2. These species facilitate oxidative reaction with TEA, as TEA molecules are adsorbed and chemically activated through interactions with Pt SAs, particularly via bonding with nitrogen atoms. This activation lowers the energy barrier for TEA oxidation, resulting in products like CO2, H2O, and NO2, which desorb efficiently, ensuring sensor repeatability and long-term stability. Moreover, Pt incorporation alters

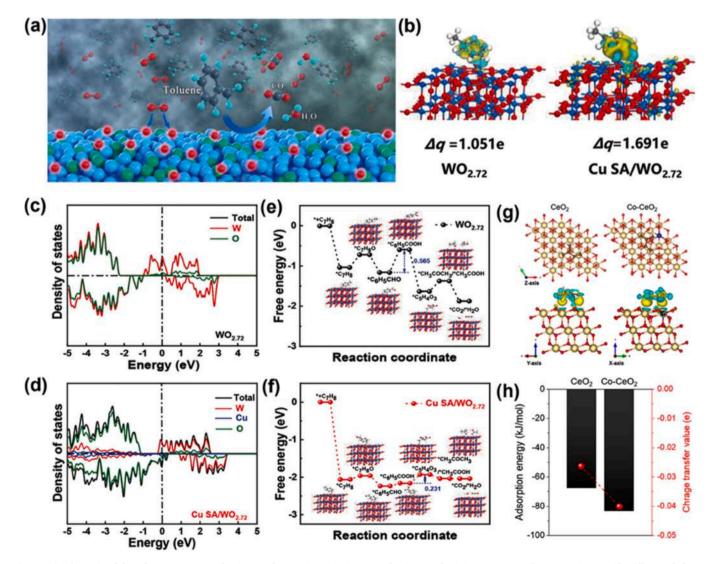


Fig. 5. (a) Schematic of the toluene sensing mechanism on the Cu SA/WO2.72 sensor. (b, c) DOS of pristine WO2.72 and Cu SA/WO2.72. (d) Differential charge density and Bader charge ( $\Delta q$ ) of toluene adsorption on both samples. (e, f) Energy pathways for toluene oxidation to CO2 and H<sub>2</sub>O on pristine WO2.72 and Cu SA/WO2.72. Reprinted with permission from Ref. [146]. (g) Optimized binding configurations and charge density differences of C5H8 adsorption on CeO2 and Co-CeO2. (h) Adsorption energy and charge transfer comparison of C5H8 on CeO2 and Co-CeO2. Reprinted with permission from Ref. [147].

the ZnO electronic structure, enhancing sensing performance. Defect level formations and a higher conduction band position (Fig. 4b-h) promote electron transfer to surface-adsorbed oxygen, increasing the concentration of reactive oxygen species. This mechanism is optimized by fine-tuning the Pt loading to preserve atomic dispersion and avoid nanoparticle aggregation that diminishes active site availability. These insights into the molecular adsorption, activation, and oxidation clarify the pathways for improved gas detection.

Transition metal SACs have also attracted attention for gas sensing, owing to their unique electronic structures, tunable d-band centers, and variable oxidation states, enabling modulation of catalytic activity for enhanced sensitivity and selectivity. Their abundance and coordination flexibility make them promising alternatives for VOC detection technologies. Further incorporating MOX to support transition metal SA is beneficial; for example, Cu sites anchored on ultrathin WO2.72 nanowires (CuSA/WO<sub>2.72</sub>) demonstrated exceptional sensing performance for toluene, with a detection limit of 10 ppb, ultrafast response and recovery times, and excellent reversibility [146] (Fig. 5a). The superior performance was attributed to synergistic effects between Cu SAs and WO2.72 nanowires. Cu sites serve as active centers for toluene adsorption, facilitating charge transfer and modulating the MOX electronic structure. Toluene adsorption induces measurable resistance changes due to enhanced catalytic decomposition, validated by in situ IR spectroscopy, which confirms selective adsorption and dynamic surface interactions. Computational studies, including DFT and MD simulations, provide atomic-level insights into enhanced adsorption energy, binding sites, and toluene stability. Upon Cu SA incorporation, electron concentration in WO<sub>2.72</sub> increased from 1.051e to 1.691e (Fig. 5b). DOS analysis (Fig. 5c, d) showed Cu SA/WO<sub>2.72</sub> with continuous occupied states at the Fermi level and a conduction band closer to it, indicating improved charge transfer. Gibbs free energy profiles for toluene oxidation (Fig. 5e, f) indicate a six-step pathway (\* $C_7H_{8^-} \rightarrow *C_7H_7-OH \rightarrow$  $^*\text{C}_6\text{H}_5-\text{CHO} \rightarrow ^*\text{C}_6\text{H}_5-\text{COOH} \rightarrow ^*\text{C}_5\text{H}_4\text{O}_3 \rightarrow ^*\text{CH}_3\text{COCH}_3 + ^*\text{CH}_3\text{COOH}$  $\rightarrow$  \*CO<sub>2</sub> + \*H<sub>2</sub>O), with Cu SAs significantly lowering the reaction energy barriers. These results validate the experimental findings and emphasize the potential of Cu SA/ WO<sub>2.72</sub> nanowires for sensitive and selective gas detection. Recent research suggests that combining MOX with other MOX phases can further enhance selectivity and sensitivity due to improved ionic and electronic charge transfer. For instance, Co-doped CeO2 (Co-CeO2) catalysts functionalized on SnO2 nanofibers significantly improved sensitivity and selectivity for isoprene (C<sub>5</sub>H<sub>8</sub>) [147]. The catalytic activity of Co-CeO2 stems from its ability to increase the density of reactive oxygen species, such as adsorbed oxygen (O<sub>2</sub>, O<sub>2</sub>) and lattice oxygen (O<sup>2</sup>-), which are crucial for gas adsorption and activation. XPS analyses reveal that the ratio of adsorbed to lattice oxygen species is substantially higher for Co-CeO2-modified SnO2 compared to pristine SnO<sub>2</sub>, facilitating stronger and more selective interactions with target gases. This enhancement is attributed to several factors: Co-CeO2 promotes the spillover of reactive oxygen species across the sensing surface, increasing available active sites. Additionally, DFT calculations indicate greater charge transfer during gas adsorption. Charge density difference and Bader charge analyses (Fig. 5g, h) reveal increased charge transfer from 0.0263e for CeO<sub>2</sub> to 0.0401e for Co – CeO<sub>2</sub>, demonstrating the role of Co doping in enhancing adsorption and activation. Co incorporation also strengthens C<sub>5</sub>H<sub>8</sub> adsorption, as evidenced by the higher adsorption energy on Co- CeO<sub>2</sub> (-83.5 kJ/mol) compared to CeO<sub>2</sub> (-67.6 kJ/mol) (Fig. 5h). These findings confirm stronger gas-surface interactions in Co-doped CeO<sub>2</sub>, reinforcing its potential for improved catalytic performance. Beyond catalytic performance, transition metal-doped MOX catalysts, such as Co-CeO2, offer a cost-effective alternative to noble metals like Pt and Pd, whose high cost and scarcity limit large-scale deployment. In contrast, transition metal SAs-doped MOXs deliver comparable benefits at lower cost, supporting their practical use in advanced chemiresistive sensors. The integration of 1D nanomaterials with SACs is further facilitated by techniques like electrospinning, which enables the fabrication of porous nanostructures with uniform atomic

dispersion. While challenges such as particle aggregation and limited nanoscale dispersion persist, using SAs-doped MOXs mitigates these issues by stabilizing reactive oxygen species and minimizing grain growth during synthesis.

Importantly, optimizing the oxygen spillover effect and tailoring interfacial f, such as electron transfer and adsorption energy modulation, significantly enhance the sensing capabilities of SAES.

# 3.2.2. 2D material supported SAE sensors

Two-dimensional (2D) nanomaterials exhibit unique properties such as large surface area, tunable band gaps, and high stability, making them highly suitable for sensing applications. As discussed in the previous section, MOS materials play a critical role in sensing mechanisms through the interplay of lattice oxygen vacancies and adsorbed oxygen species, optimizing the oxygen spillover effect depending on the type of VOC. The transformation of semiconducting MOS into 2D structures has rendered them particularly attractive for chemiresistive sensors due to their enhanced active surface area, low power consumption, excellent charge carrier mobility, high modifiability, and scalability.

Recent advancements in 2D MOS-supported SAES have revolutionized gas detection technologies. For example, WO3, an n-type semiconductor with a narrow bandgap and superior charge transport properties, is widely employed in gas sensors. Functionalization of defective WO<sub>3</sub> nanoplates with single Pt atoms (SA-Pt@WO<sub>3-x</sub>) has significantly enhanced sensing capabilities, particularly toward acetone detection [108]. The mechanism leverages defect sites on WO<sub>3-x</sub> to anchor Pt SAs, forming highly active catalytic sites that facilitate oxygen adsorption and redox reactions. Adsorbed oxygen species extract electrons from the conduction band to form oxygen ions, resulting in the formation of an electron depletion layer (Fig. 6a). Upon exposure to acetone, redox reactions release electrons back into the conduction band, causing a measurable decrease in resistance. This process is accompanied by the formation of CO<sub>2</sub> and H<sub>2</sub>O, as depicted in Fig. 6b. The isolated Pt SAs lower activation energy barriers, enhance electron transfer, and maintain uniform coordination, delivering exceptional

sensitivity ( $R_{air}/R_{gas}=43.54$  for 5 ppm acetone), selectivity, and ultralow detection limits (down to 0.1 ppm). The SA-Pt@WO<sub>3-x</sub> platform exhibits rapid response and recovery times (7 s and 24 s, respectively), along with excellent reproducibility and long-term operational stability for up to 49 days.

In another study, SAs of Au on In<sub>2</sub>O<sub>3</sub> nanosheets were employed to develop a highly sensitive method and selective for HCHO detection [105]. This design demonstrated that a minimal Au loading of 0.25 % significantly improved sensing performance, achieving an exceptional response (85.67) to 50 ppm HCHO at a low operating temperature (100 °C) with a detection limit of 1.42 ppb. The synthesis involved hydrothermal preparation of In<sub>2</sub>O<sub>3</sub> nanosheets, followed by calcination at 300 °C. Au single atoms were uniformly loaded via UV irradiation of an HAuCl<sub>4</sub> solution. Mechanistically, Au SAs acts as electron sinks, promoting the adsorption of oxygen molecules and catalyzing redox reactions with HCHO to form CO2 and H2O (Fig. 6c). This interaction reduces the activation energy (Ea = 4.7 kJ/mol for  $0.25 \% \text{ Au/In}_2\text{O}_3$ ), enhances charge transfer efficiency, and provides uniformly distributed active sites for selective HCHO adsorption. The sensor also demonstrates high selectivity, long-term stability over 30 days, and excellent resistance to humidity in the 20-40 % RH range. This study highlights the critical role of Au SAs in tuning the surface chemistry of In2O3, improving sensitivity, and reducing the operating temperature, making it a promising candidate for practical applications in HCHO detection. 2D material-supported SA sensors represent a significant leap forward in VOC sensing, offering a combination of high sensitivity and robust operational stability. While SACs on 2D materials enable precise atomic-level modulation of adsorption and electron transfer, MOSs contribute high stability, cost-effectiveness, and efficient surface reactivity, together enabling highly efficient VOC detection.

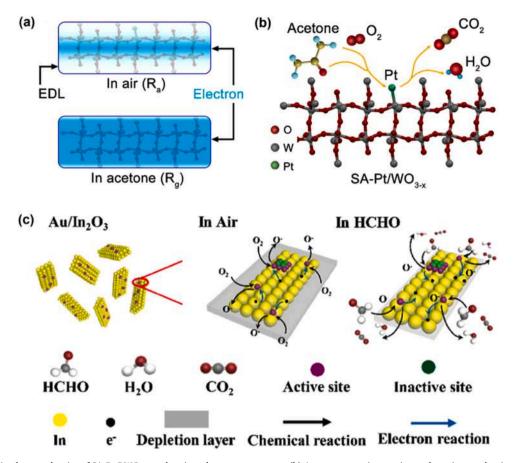


Fig. 6. (a) Variation in electron density of SA-Pt@WO<sub>3-x</sub> under air and acetone exposure. (b) Acetone vapor interaction and sensing mechanism on the SA-Pt@WO<sub>3-x</sub> surface. Reprinted with permission from Ref. [108]; (c) Schematic representation of the HCHO detection process on  $Au@In_2O_3$ . Reprinted with permission from Ref. [105].

# 3.2.3. 3D material-based SAE sensors

Beyond mesoporous oxides, three-dimensional (3D) MOS, metalorganic frameworks (MOFs), and conductive CMOFs have emerged as promising hosts for SAs. These porous architectures provide high surface areas, tunable pore structures, and abundant active sites, which significantly enhance oxygen adsorption. This increased oxygen uptake, in turn, facilitates greater electron depletion layer formation, leading to a stronger sensor response in VOC detection.

A novel chemiresistive gas sensor was developed by incorporating Pt-SAs into an Ag-LaFeO3@ZnO core-shell structured material using a molecular imprinting-assisted sol-gel method [148]. The Pt-SAs were dispersed within a ZnO framework derived from the MOF precursor ZIF-8, ensuring strong Pt-O-Zn interactions that enhanced catalytic activity. The resulting Ag-LaFeO3@ZnO-Pt sensor exhibited an exceptionally high specific surface area of 192.08 m<sup>2</sup>/g, significantly improving the working temperature and detection limit for methanol. The sensor exhibited ultrahigh sensitivity of 453.02-5 ppm CH<sub>3</sub>OH at 86 °C and retained a strong response even at 62 ppb (21.25). Compared to the Ag-LaFeO3@ZnO sensor, the Pt-modified version exhibited a 6.69-fold increase in sensitivity, with a minimum detection limit of 3.27 ppb. DFT calculations revealed that the unoccupied 5d states of Pt SAs significantly enhanced oxygen and methanol adsorption, leading to stronger electron depletion effects and improved redox reaction kinetics. The methanol oxidation process on Ag-LaFeO<sub>3</sub>@ZnO-Pt facilitated the formation of \*CH<sub>3</sub>O, \*CH<sub>2</sub>O, and \*CO<sub>2</sub>, with lower activation energy barriers compared to its non-Pt-counterpart, as shown in Fig. 7a. These findings confirmed that Pt SAs play a crucial role in boosting methanol sensing performance by promoting charge transfer and surface reaction efficiency.

Building on these findings, another study used a MOF-derived N-

doped graphene (NG) framework to anchor atomically dispersed Pt SAs on  $\rm In_2O_3$  via a sacrificial templating route [73]. This synthesis approach integrates an MOF precursor with an NG template, effectively tuning the physicochemical properties of  $\rm In_2O_3$  and stabilizing single-atom Pt on its surface. The inclusion of Pt SACs increases the specific surface area, oxygen vacancies, and adsorbed oxygen species,

providing more active sites for gas adsorption. As a result, the Pt-In<sub>2</sub>O<sub>3</sub>-NG sensor exhibits a high response (750.4–100 ppm), rapid response time (2 s to 100 ppm), excellent selectivity, and a low theoretical detection limit (8.4 ppb). As shown in Fig. 7b, the sensing mechanism is based on electron transfer between HCHO molecules and chemisorbed oxygen species. Oxygen molecules adsorb onto the In<sub>2</sub>O<sub>3</sub> surface, forming O<sub>2</sub>-, O-, and O<sub>2</sub>- species, which create an electron depletion layer, increasing resistance.

Upon HCHO exposure, oxidation reactions release electrons, reducing resistance and enabling detection. The  $Pt-In_2O_3-NG$  exhibits the highest concentration of active oxygen species participating in the sensing process. The introduction of Pt SAs also modifies the electronic structure of  $In_2O_3$ , reducing its band gap and facilitating stronger interactions between Pt and Pt HCHO molecules, thus improving sensor performance. By leveraging the tunable porosity and electronic properties of MOF-derived structures, this work demonstrates the effectiveness of single-atom catalysts in enhancing the sensitivity, selectivity, and operational efficiency of chemiresistive gas sensors.

Further, synthesis strategies and the choice of cocatalysts play a pivotal role in determining the gas-sensing performance of metal oxide-based sensors. The structural and electronic modifications induced by different fabrication techniques can significantly influence the density of active sites, oxygen vacancy concentration, and charge transport properties, ultimately affecting sensitivity and selectivity. To illustrate this,

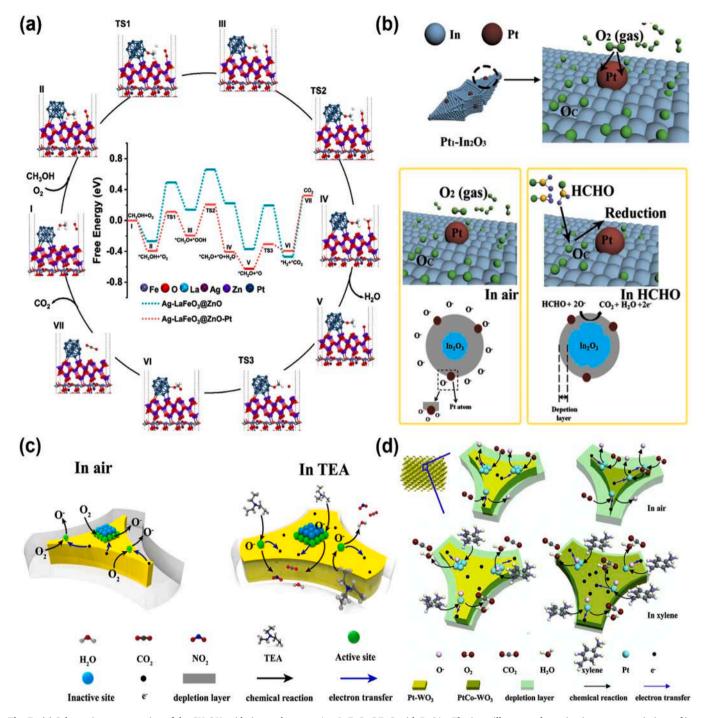


Fig. 7. (a) Schematic representation of the  $CH_3OH$  oxidation pathway on Ag—LaFeO<sub>3</sub>@ZnO with Pt SAs. The inset illustrates the activation energy variations of ions and molecules in Ag—LaFeO<sub>3</sub>@ZnO and Ag—LaFeO<sub>3</sub>@ZnO—Pt at different reaction stages. The reaction cycle highlights the key intermediates and transition states (TSs). Reprinted with permission from Ref. [148]. (b) Conceptual depiction of the gas sensing mechanism in Pt-In<sub>2</sub>O<sub>3</sub> [73]. (c) Mechanistic illustration of TEA detection and conversion on Pt@WO<sub>3</sub> Reprinted with permission from Ref. [149]. (d) Diagrammatic representation of the xylene sensing process in Pt@WIn<sub>2</sub>O<sub>3</sub> and PtCo-WIn<sub>2</sub>O<sub>3</sub> systems. Reprinted with permission from Ref. [150].

case studies on  $WO_3$  coordinated with SAs through different cocatalyst choices and synthesis approaches have been explored. Gu et al. conducted a series of studies on 3D Pt SAs and Pt combined with transition metal SAs on  $WO_3$  for detecting trimethylamine (TMA) and xylene, employing the colloidal crystal template method with a PMMA sacrificial template [149,150]. Their findings revealed that, compared to Pt- $WO_3$  nanoparticles, Pt SAs@ $WO_3$  exhibited significantly enhanced sensitivity of 28.37 ppm and an ultralow LOD of 0.18 ppb for TMA. This improvement was attributed to reduced activation energy, which

facilitates higher oxygen adsorption and improved gas-sensing performance. Additionally, the authors investigated the influence of Ni and Co transition metal SAs as cocatalysts in the 3D Pt SA-WO $_3$  structure. Among the studied materials, PtCo-SA@WO $_3$  demonstrated a superior sensitivity of 3.91 ppm and an LOD of 1.08 ppb for xylene detection. The incorporation of transition metals further increased oxygen vacancy concentration, leading to enhanced Pt SAs dispersion, a reduced band gap, and improved electron mobility within WO $_3$ . The uniform distribution of Pt SAs played a crucial role in increasing the density of active

sites and the adsorption capacity for oxygen molecules. As shown in Fig. 7c and 7d, the heightened oxygen adsorption expanded the electron depletion layer, resulting in significant resistance variations in the sensing material and consequently improving gas-sensing performance.

Expanding upon this, Dai et al. employed a wet impregnation method to anchor Pt SAs onto WO3, leveraging surface oxygen vacancies for stabilization [71]. To enhance WO3-based chemiresistors, they combined chemical sensitization via Pt SAs with physical sensitization through pulsed temperature modulation (PTM). The incorporation of Pt SAs significantly improved sensor response to TMA and xylene, with 0.1 at% Pt SAs anchoring increasing response values from 2.7 to 7.8 for TMA and from 2.1 to 5.4 for xylene. PTM further amplified these

responses to 6541.5 and 1001.1, respectively, achieving ultra-high sensitivity with theoretical detection limits as low as 0.78 ppt for TMA and 0.18 ppt for xylene. This study delves into the gas-sensing mechanism of Pt SAs on WO<sub>3</sub>, emphasizing the role of surface chemistry and dynamic oxygen species in enhancing chemiresistive performance. Pt SAs modulate the electronic structure of WO<sub>3</sub>, thereby improving VOC adsorption and catalytic activation. The enhancement is attributed to dynamic oxygen species modulation and improved analyte diffusion, facilitated by PTM. Mechanistically, Pt SAs promote the formation and replenishment of active oxygen species (O<sub>2</sub>¯, O¯), which are crucial for VOC oxidation. Under PTM, periodic high-temperature pulses regenerate these oxygen species, preventing sensor saturation and

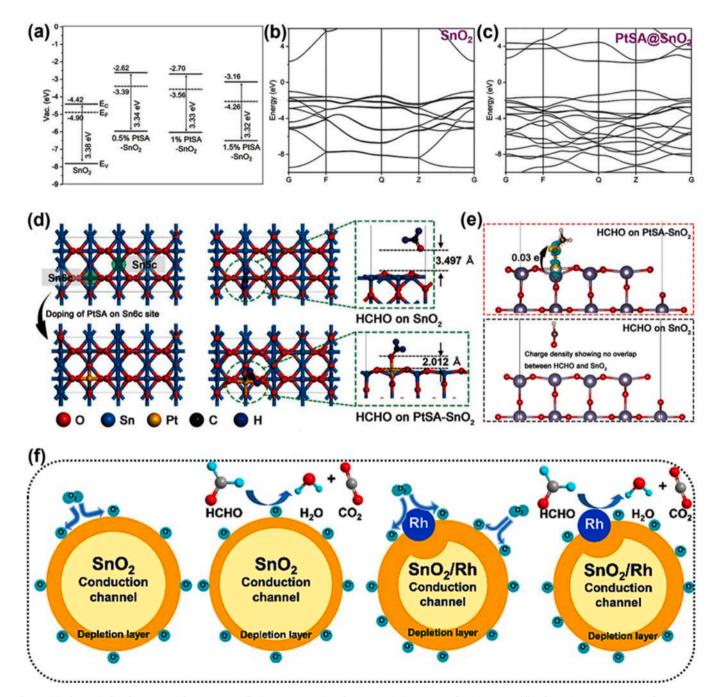


Fig. 8. (a) Electronic band structure of PtSA@SnO<sub>2</sub>. (b-c) DOS comparison between pristine SnO<sub>2</sub> and PtSA@SnO<sub>2</sub>. (d) Schematic representation of PtSA@SnO<sub>2</sub> surface formation, along with the most stable configurations of HCHO adsorption on SnO<sub>2</sub> and PtSA@SnO<sub>2</sub> surfaces. (e) Charge density difference analysis for HCHO adsorption on SnO<sub>2</sub> and PtSA@SnO<sub>2</sub>. Reprinted with permission from Ref. [151]. (f) Illustration of the gas sensing mechanism of SnO<sub>2</sub> and RhSA-SnO<sub>2</sub> under ambient conditions and formaldehyde exposure. Reprinted with permission from Ref. [74].

accelerating response/recovery kinetics. This dynamic modulation significantly enhances sensing performance, enabling ultra-low detection limits. Furthermore, Pt SAs serve as catalytic sites, lowering the activation energy for VOC oxidation while ensuring long-term sensor stability. PTM further optimizes gas diffusion and surface reactions by periodically desorbing surface-bound intermediates, maintaining high sensitivity over multiple sensing cycles. These findings offer a comprehensive perspective on how the choice of synthesis methods and co-catalysts impacts the sensitivity of the materials.

As discussed in the 1D SAs sensors section, in the detection of HCHO, SnO2 is well reported due to its selectivity towards HCHO. A 3D Pt SAdecorated SnO2 material was synthesized using a carbonization-oxidation strategy combined with ultrasonic spray pyrolysis to obtain Pt SA substitutionally doped SnO<sub>2</sub> (PtSA@SnO<sub>2</sub>) [151]. The structural analysis confirms that Pt SAs are embedded within the SnO<sub>2</sub> lattice, modifying its electrical properties and band structure, resulting in a low-resistivity state (Fig. 8a-c). The PtSA@SnO2 sensor enables humidity-independent detection of trace-level HCHO (10 ppb) with high response, fast response speed, and excellent selectivity, even under 70 % RH conditions. Unlike conventional sensors, increasing Pt SA doping decreases resistance, which benefits circuit integration. DFT calculations revealed that Pt SA doping narrows the band gap (from 3.34 eV in SnO<sub>2</sub>) to 3.08 eV in PtSA@SnO2), while the Pt 5d orbital introduces new electronic states, enhancing electron transfer and conductivity (Fig. 8a). HCHO adsorption energy on PtSA@SnO2 is significantly higher (-1.123 eV vs. -0.327 eV for pure SnO₂), indicating stronger gas interactions (Fig. 8d). As depicted in Fig. 8e, the presence of Pt SA significantly boosts charge transfer, improving both sensor selectivity and sensitivity. The PtSA@SnO2 sensor demonstrates ultra-low detection limits (10 ppb), a rapid response time (24 s), excellent selectivity, and long-term stability.

Rh SAs have been employed to enhance the gas-sensing performance of SnO<sub>2</sub> sensors through atomic layer deposition (ALD) [74]. Gas-sensing investigations reveal that RhSA-SnO<sub>2</sub> exhibits a remarkable response of 36.3-20 ppm formaldehyde, representing a nearly 23-fold improvement over pure SnO2. This sensor also demonstrates high sensitivity and rapid response-recovery times (4-29 s). Optimizing the Rh SA loading, RhSA-SnO<sub>2</sub> with 25 ALD cycles of Rh achieves the highest response to HCHO at 250 °C, further confirming the 23-fold enhancement compared to pristine SnO2. Additionally, RhSA-SnO2 maintains outstanding selectivity and stability, with an ultralow detection limit of 55 ppb. The superior sensing performance of Rh-doped SnO<sub>2</sub> is attributed to enhanced HCHO adsorption and charge transfer, as illustrated in Fig. 8f. The interaction energy between HCHO and RhSA-SnO<sub>2</sub> (-1.99 eV) is significantly higher than that of pure SnO<sub>2</sub> (-1.69 eV), leading to stronger analyte binding. Furthermore, charge transfer is greatly increased (2.27 e vs. 0.125 e), substantially improving sensor sensitivity. Selectivity for HCHO over other gases is also enhanced due to the stronger interaction energy. Density of states analysis confirms that Rh acts as the primary active site, with Rh-O interactions playing a pivotal role in adsorption. Moreover, Rh doping facilitates oxygen adsorption and desorption, increasing the surface oxygen content, which contributes to superior sensing performance. Hybridization between O-p and Rh-d orbitals is identified as the key factor driving the improved HCHO response in RhSA-SnO2.

These findings collectively underscore the significance of 3D SA@MOS-based VOC sensors in achieving enhanced sensitivity, selectivity, and stability. The advancements in 3D SA-based materials integrated into MOS have significantly enhanced the performance of VOC sensors. These architectures provide higher surface areas, tunable pore structures, and abundant active sites, facilitating superior oxygen adsorption and improving electron depletion layer formation. The incorporation of single-atom catalysts, such as Pt and Rh, has demonstrated remarkable improvements in sensor sensitivity, selectivity, and operational stability by modulating electronic structures and catalytic activity. Notably, studies on Pt-SA-modified ZnO, In<sub>2</sub>O<sub>3</sub>, and WO<sub>3</sub> have

revealed enhanced gas adsorption, charge transfer efficiency, and reduced activation energy barriers, leading to ultralow detection limits and rapid response times. The interplay between SACs and transition metal cocatalysts, such as Ni and Co, has further optimized oxygen vacancy concentration and electronic properties, improving the overall sensing mechanism. Additionally, the application of PTM in WO<sub>3</sub>-based sensors has been pivotal in regenerating active oxygen species, sustaining sensor performance over multiple cycles. The introduction of Pt and Rh SAs in SnO<sub>2</sub>-based sensors has demonstrated a profound impact on HCHO detection, with significant band gap reduction, enhanced charge density, and improved humidity tolerance. The synergistic effect of SAs and host materials underlines the importance of rational material design in developing next-generation chemiresistive VOC sensors. These findings collectively highlight the transformative role of 3D SA@MOSbased sensors in advancing gas-sensing technologies, emphasizing the critical influence of synthesis strategies, cocatalyst selection, and electronic structure engineering in achieving superior sensor performance.

### 3.3. Electrochemical SAE sensors

Electrochemical sensors based on SAs have emerged as powerful tools for detecting environmental pollutants with high sensitivity and selectivity. The unique electronic structure and well-defined active sites of SACs facilitate efficient electron transfer and catalytic activity, making them ideal candidates for electrochemical sensing applications. Incorporating SAs into conductive support materials enhances sensor performance by improving charge transport, increasing active surface area, and tailoring adsorption properties. Transition metal SAs have received considerable attention for their potential in gas sensing applications. Their unique electronic configuration and variable oxidation states offer opportunities to fine-tune catalytic sites, improving both sensitivity and selectivity. Their abundance and versatile coordination chemistry further establish them as promising candidates for advancing VOC detection technologies.

A notable example is the Nb@BNT sensor for nitrobenzene (NB) detection, reported by Li et al. synthesized via a co-precipitation method followed by high-temperature pyrolysis [124]. The SA Nb-BCN-modified glassy carbon electrode (GCE) exhibited a superior electrochemical response, underscoring the critical role of SACs in pollutant sensing. Mechanistic studies revealed that the incorporation of isolated Nb atoms into the boron-carbon-nitrogen (BCN) framework generated highly active sites that enhanced catalytic activity and enabled selective adsorption of Nb molecules. The synergistic interaction between Nb SAs and the surrounding matrix significantly lowered the energy barrier for Nb reduction, evidenced by a positive shift in reduction peak potential (0.6 V) and a remarkable 42-fold increase in peak current compared to the bare GCE. Further investigations into the catalytic mechanism confirmed that Nb SAs played a crucial role in stabilizing reaction intermediates and facilitating a multi-electron transfer process for Nb reduction. The electrochemical behavior of the sensor was systematically optimized by tuning parameters such as enrichment time, applied potential, and pH, leading to an ultralow detection limit of 0.70 mM. Additionally, the sensor exhibited dual linear response ranges for different NB concentrations, enhancing its adaptability to diverse environmental conditions. Real-world validation was performed through NB detection in tap and lake water samples, where high recovery rates confirmed the practical applicability of SAES in environmental monitoring. These advancements demonstrate the remarkable potential of SA-based electrochemical sensors in VOC detection, particularly through improvements in selectivity, sensitivity, and stability under complex and fluctuating environmental conditions. Notably, the ability of these sensors to maintain robust performance across a wide pH range is particularly crucial for real-world applications, given the variable acidity and alkalinity of environmental and industrial samples. Despite their promising attributes, SA-based electrochemical sensors remain relatively underexplored, and further investigations could unlock their full potential in gas sensing technologies. Additionally, DFT studies could provide deeper mechanistic insights into the electronic structure, adsorption behavior, and reaction pathways of SA sites, thereby enabling the rational design of electrochemical sensors with improved sensitivity, selectivity, and stability. Such innovative approaches will drive the development of next-generation electrochemical sensors with superior durability, selectivity, and real-world applicability in environmental monitoring and beyond.

### 3.4. SAE Field effect transistor sensors

FET sensors, including metal-oxide-semiconductor FET (MOSFET), thin-film transistor (TFT), and chemical field-effect transistor (Chem-FET), function by modulating the conductivity of a semiconducting channel upon exposure to analyte gases. This modulation arises from charge transfer or dipole interactions between the gas molecules and the channel material, imparting high sensitivity, rapid response/recovery times, and room temperature operability. These features render FETs well-suited for detecting a wide range of gases, including VOCs (Table 3). However, conventional FET sensors are often limited by poor selectivity, limited long-term stability, and performance degradation under humid conditions (Table 4) [152-157]. To overcome these limitations, MXenes have emerged as promising channel materials due to their metallic conductivity, tunable surface terminations, and large specific surface area, which provide abundant active sites for gas adsorption. Additionally, MXenes offer excellent platforms for stabilizing SACs, owing to their ability to anchor metal adatoms effectively. Incorporating SAs onto MXenes substrates introduces highly reactive catalytic sites that significantly enhance gas adsorption and charge

**Table 3**Comparison of FET architectures and their suitability for VOC sensing.

S. No	Parameter	MOSFET (Metal- oxide- semiconductor FET)	TFT (Thin-film transistor)	ChemFET (Chemical field-effect transistor)
1	Structure	Bulk or planar silicon-based architecture	Uses thin-film semiconductors on substrates	Modified FET with chemically sensitive gate
2	Material Flexibility	Low (mainly Sibased)	High (can use organic/inorganic materials)	High (gate surface can be functionalized with receptors)
3	Sensitivity to VOCs	Limited (requires surface functionalization)	Moderate to high (depending on material)	High (direct chemical interaction with gate)
4	Selectivity	Low (without modification)	Moderate (enhanced by nanomaterials)	High (with specific receptors or SAMs)
5	Response Time	Fast	Moderate to fast	Fast to moderate (depends on diffusion)
6	Operating Voltage	Moderate to high	Low to moderate	Low
7	Fabrication Cost	High (CMOS processes)	Low (solution- processable materials possible)	Moderate
8	Integration with Flexible Substrates	Poor	Excellent	Good
9	Scalability	Excellent (mature CMOS technology)	Good	Moderate
10	Real-world Applicability	Limited in VOC sensing	Suitable for low- cost, flexible sensors	Highly suitable for selective VOC sensing

transfer processes.

A recent study reported a FET sensor based on Pt SAs anchored onto ultrathin Ti<sub>3</sub>C<sub>2</sub>T<sub>x</sub> nanosheets for detecting TEA gas at room temperature [70]. Pristine Ti<sub>3</sub>C<sub>2</sub>T<sub>x</sub> nanosheets were obtained through liquid-phase exfoliation and lithium fluoride/hydrochloric acid etching, generating Ti vacancies that acted as anchoring sites for Pt SAs. The Pt SAs were immobilized via self-reduction of Pt4+ under argon agitation, with Ti<sub>3</sub>C<sub>2</sub>T<sub>x</sub> acting as the reductant. Introducing Pt SAs lowered the activation energy for gas adsorption and improved selectivity toward TEA. Mechanistically, Pt SAs catalyzed the dissociation of oxygen molecules through the spillover effect, generating active O species that oxidized TEA into NO<sub>2</sub>, CO<sub>2</sub>, and H<sub>2</sub>O (Fig. 9c). This reaction released electrons into the Ti<sub>3</sub>C<sub>2</sub>T<sub>x</sub> matrix, reducing the hole density in the p-type semiconductor and thereby decreasing its conductivity. This sensor achieved an impressive detection limit of 14 ppb, rapid response times of 3 s at 1 ppm TEA, and excellent ambient stability. The study also investigated the adsorption energy of Pt SAs with various MXene terminations (-OH, -O-, and -F) for TEA, revealing that Pt SA-MXene with F-Pt sites exhibited the highest TEA binding affinity (Fig. 9d, e). Beyond MXenes, TMDs such as MoS2, WS2, and MoSe2, stand out due to their semiconducting nature and direct band gaps, which enable effective FET-based sensing. Modifying 2D materials via doping or heterostructure formation further enhances their gas sensing capabilities by improving adsorption and catalytic activity. In this context, the incorporation of SAs into chalcogenide atom vacancies in TMDs strengthens their interaction with VOC molecules. One study demonstrated that functionalizing 2D materials (n-type MoS<sub>2</sub> and p-type WSe<sub>2</sub>) with SAs (Pt, Co, Ru) significantly enhanced VOC detection performance. Notably, even a minimal Pt loading (0.1 %) on MoS<sub>2</sub> yielded a high sensitivity of 92.3 for detecting 50 ppm acetone at room temperature (25 °C), with improved selectivity over ethanol and water [159]. This sensor also demonstrated an impressive detection limit of 0.05 ppm, highlighting its capability for low-concentration detection.

The Pt@MoS2 sensor was synthesized via a one-step UV light-assisted reduction process, wherein Pt SAs were deposited onto mechanically exfoliated MoS<sub>2</sub>. Mechanistically, the n-type nature of MoS<sub>2</sub> facilitated electron transfer upon VOC adsorption, increasing surface electron density at the surface and lowering the activation energy (Ea = 23.5 kJ/mol for Pt/MoS<sub>2</sub>), which enhanced both adsorption and desorption kinetics. Conversely, the p-type WSe2 stabilized adsorbed VOC molecules through charge delocalization and selective binding (Fig. 10a), enabling efficient detection and recovery. These complementary properties of TMDs, combined with SAs decoration, optimize adsorption capacity, charge transfer, and selectivity. The sensor exhibited excellent selectivity toward acetone, operational stability for over 50 days, and robust performance under high humidity (30-60 % RH). These findings highlight the transformative role of SA catalysts in tuning the electronic properties of n-type and p-type TMD substrates, facilitating dynamic adsorption-desorption processes under variable environmental conditions. The synergy between n-type and p-type 2D materials enables high sensitivity, selectivity, and efficient low-temperature operation, making these systems promising candidates for environmental monitoring, industrial safety, low power consumption, and healthcare diagnostics.

In summary, the integration of SAs on MXenes and TMDs significantly enhances the sensing performance of FET-based sensors by optimizing gas adsorption kinetics and catalytic activity. This advancement underscores the versatility and potential of SA-decorated MXenes and TMDs for developing next-generation VOC sensors with ultrahigh sensitivity, selectivity, stability, and room temperature operation. Continued exploration of various SAs and support materials is expected to further expand their applicability across diverse sensing platforms.

### 3.5. Microelectromechanical system (MEMS)

SA-incorporated materials have emerged as a promising class of catalysts for enhancing the sensitivity, selectivity, and response time of

**Table 4**Comparison of SA FETs and conventional nanomaterial-based FET sensors for VOC detection.

S.No	FET Type	Catalyst	Target VOC	<b>Detection limit</b>	Response time	Operating temperature (°C)	Ref
1	SA-FET	Pt SA-Ti <sub>3</sub> C <sub>2</sub> T <sub>x</sub>	Triethylamine	14 ppb	5 s	Room Temp.	[70]
2	NP-FET	Pd nanoparticles on CNT	Benzene	~50 ppb	8 s	Room Temp.	[152]
3	NP-FET	SnO <sub>2</sub> mesoporous nanoparticles	Formaldehyde	~10 ppm	~15 s	~200 °C	[153]
4	s-CNTs-FET	s-CNTs	Formaldehyde	20 ppb	-	Room Temp.	[154]
5	NP-FET	RGO/ZnO	Formaldehyde	120 ppb	120 s	Room Temp.	[155]
6	NP-FET	1T-2H-MoS <sub>2</sub>	Formaldehyde	1 ppm	150 s	Room Temp.	[156]
7	NP-FET	ZnO/WS <sub>2</sub>	Benzene	500 ppb	90 s	Room Temp.	[157]

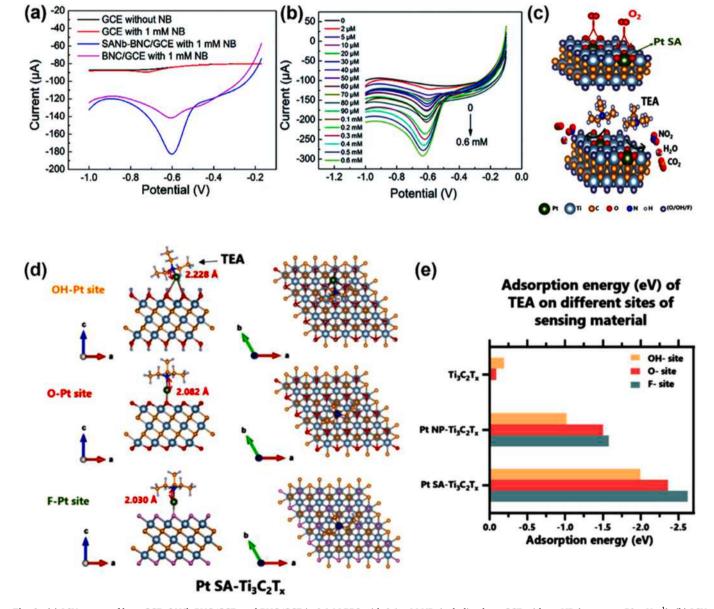


Fig. 9. (a) LSV curves of bare GCE, SANb-BNC/GCE, and BNC/GCE in 0.1 M PBS with 0.1 mM NB, including bare GCE without NB (scan rate: 50 mV  $s^{-1}$ ). (b) LSV curves at varying NB concentrations (0-600  $\mu$ M) under optimized conditions (pH 7.0, 0.1 M PBS). Reprinted with permission from Ref. [124]. (c) Schematic of Pt SA-facilitated TEA adsorption and surface reaction on Pt SA@Ti\_3C\_2T\_x. (d, e) DFT models and adsorption energies of TEA with N-terminal atoms on OH , O , and F sites of Pt SA@Ti\_3C\_2T\_x, Pt NP@Ti\_3C\_2T\_x, and pristine Ti\_3C\_2T\_xReprinted with permission from Ref. [70].

MEMS sensors in VOC detection. The superior catalytic activity of these materials arises from atomically dispersed active metal sites, which provide abundant active centers, tunable surface interactions, and enhanced gas adsorption capabilities. These attributes facilitate efficient charge transfer and oxidation processes, enabling selective VOC detection even at low concentrations and reduced operating temperatures.

A representative case involves the detection of 3-hydroxy-2-butanone (3H-2B), a key biomarker VOC associated with *Listeria monocytogenes*, which can be selectively adsorbed and catalytically oxidized on SA sites, generating distinct and measurable signals. In this study Au-SAs were anchored on mesoporous SnO<sub>2</sub> nanospheres via a polyphenolassisted assembly of glutathione-modified Au nanoclusters, followed by

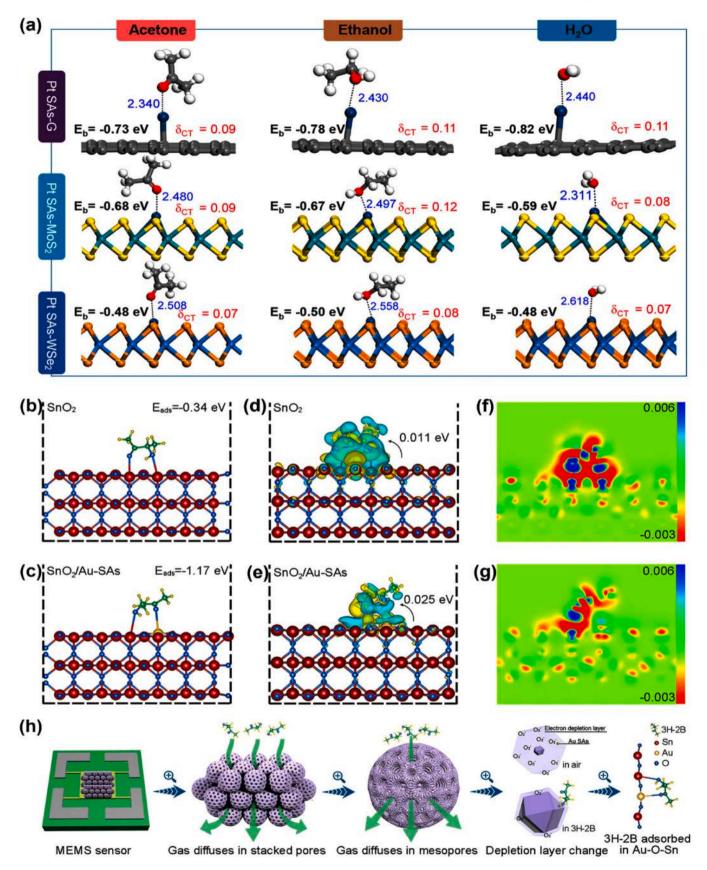


Fig. 10. (a) DFT-optimized structures of Pt SA-decorated graphene/MoS<sub>2</sub>/WSe<sub>2</sub> with adsorbed VOCs (acetone, ethanol, and water), showing DM-O distances, binding energies (Eb, eV), and charge transfer ( $\delta_{CT}$ , e). Reprinted with permission from Ref. [158]. Adsorption energies for 3H-2B on (b) SnO<sub>2</sub> and (c) SnO<sub>2</sub>/Au-SAs. Charge distributions for (d) SnO<sub>2</sub> and (e) SnO<sub>2</sub>/Au-SAs, and charge density differences in the 2D plane for (f) SnO<sub>2</sub> and (g) SnO<sub>2</sub>/Au-SAs. (h) Schematic of SnO<sub>2</sub>/Au-SAs enhanced performance for 3H-2B detection. Reprinted with permission from Ref. [159].

thermal calcination to generate atomically dispersed Au sites [158]. The resulting SA-functionalized sensor exhibited a remarkably high response (587.3) to 2 ppm of 3H-2B at just 50°C, with a fast response time of 10 s and an ultra-low detection limit of 10 ppb. Compared to conventional SnO2-based sensors, this SA-based material not only enhanced sensitivity by a factor of 183.5 but also significantly lowered the operational temperature from 100 °C to 50 °C, making it highly energy-efficient. The superior sensing performance of SnO<sub>2</sub>/Au-SA material was attributed to electronic and chemical sensitization mechanisms. These sensors were evaluated under air and nitrogen atmospheres using 2 ppm of 3H-2B as the target analyte. In a nitrogen atmosphere, the sensor exhibited a response of 300.3 with a resistance of 5.51 G $\Omega$ , while in air, the response significantly increased to 587.3. This notable enhancement suggests that Au SAs promote the adsorption and dissociation of O2, increasing the concentration of ionized oxygen species, which amplifies the response. The high response observed even in nitrogen (300.3) indicates that both electronic and chemical sensitization mechanisms play critical roles in improving sensing efficiency. Mechanistically, Au has a higher work function than SnO<sub>2</sub>, the functionalization of SnO<sub>2</sub> with Au SAs facilitating electron transfer from SnO<sub>2</sub> to Au until Fermi level equilibrium is achieved. This process increases the depletion layer thickness and electrical resistance, thereby intensifying the response upon exposure to 3H-2B vapor. Computational analysis further confirms this enhancement, with the adsorption energy increasing from -0.34 eV for pristine SnO<sub>2</sub> to -1.17 eV for AuSAs@SnO<sub>2</sub> (Fig. 10 b, c). Charge density difference analysis (Fig. 10 d-g) showed that the introduction of Au SA increased the local electron density from 0.011 to 0.022 eV, enhancing electron mobility. The synergistic effects of atomic-scale active sites, mesoporous architecture, and optimized charge transport and fast oxygen adsorption/dissociation kinetics significantly improved VOC detection (Fig. 10h).

Moreover, the significantly reduced energy demands of SA catalysts align perfectly with the low-power requirements of MEMS devices, making them highly suitable for portable and wearable applications. By integrating SA-functionalized materials into MEMS-based sensors, researchers can achieve high-performance VOC detection with minimal energy consumption, paving the way for next-generation smart sensing technologies (Table 5) [160].

# 4. Mechanistic insights from in situ, theoretical and machine learning studies

Understanding the synergistic interactions between SACs and target analytes is crucial for optimizing their performance in VOC sensing and catalytic applications. Advanced spectroscopic techniques, DFT calculations, and ML models offer significant mechanistic insights into these interactions.

#### 4.1. In Situ and operando spectroscopic investigations

In situ and operando spectroscopic techniques play a pivotal role in the real-time monitoring of catalytic and sensing processes, providing crucial insights into reaction intermediates, pathways, active sites, and variations in electronic structures. These techniques facilitate a deeper understanding of catalytic mechanisms by enabling direct observation of dynamic transformations. Key spectroscopic techniques include diffuse reflectance infrared Fourier transform spectroscopy (DRIFTS), Raman spectroscopy, X-ray absorption near-edge structure (XANES), and extended X-ray absorption fine structure (EXAFS).

# 4.1.1. Diffuse reflectance infrared Fourier transform Spectroscopy (DRIFTS)

DRIFTS is a highly effective technique for investigating surface interactions in catalysts, particularly SACs. By analyzing the adsorption and transformation of probe molecules such as CO and VOCs, DRIFTS yields valuable insights into the electronic states, coordination environments, and reaction pathways of active sites.

4.1.1.1. Differentiating metal species via CO-DRIFTS. CO-DRIFTS is instrumental in distinguishing different metal species based on their electronic structures. Several studies have demonstrated the effectiveness of CO-DRIFTS in distinguishing different SA species based on their characteristic absorption peaks. For instance, specific CO absorption peaks at 2143 cm $^{-1}$  (Pt $^{+n}$ ), 2112 cm $^{-1}$  (Pt $^{0}$  isolated), and ~2094 cm $^{-1}$ (Pt metallic clusters) represent oxidized, isolated, and metallic Pt species, respectively (Fig. 11a-c) [150]. The sharp and intense 2112 cm<sup>-1</sup> peak observed in SA-Pt/WO3 indicates a high density of isolated Pt atoms. Furthermore, changes in the full width at half maximum (FWHM) of these peaks provide information about the size distribution and dispersion of Pt species. Further investigations using DRIFTS have demonstrated how the introduction of transition metal co-catalysts influences the anchoring of SA sites. In a representative example, incorporation of Co and Ni into the Pt-WO3 system alters CO adsorption behavior, indicating shifts in the Pt oxidation state. An increase in oxygen vacancies improves the anchoring and stability of SA-Pt sites, enhancing catalytic activity. In another case study on Au/In<sub>2</sub>O<sub>3</sub> catalysts, CO peaks observed at 2112 cm<sup>-1</sup> (Au-CO) and 2172 cm<sup>-1</sup> (In<sub>2</sub>O<sub>3</sub>-CO) reveal details about the oxidation state and electronic interactions of Au atoms [105]. A shift in the Au-related peak with decreasing Au content indicates variations in oxidation cluster formation. The correlation of CO-DRIFTS data with XPS and HAADF-STEM further validates the atomic-level dispersion of Au atoms.

4.1.1.2. DRIFTS in understanding SA-based gas sensing mechanisms. DRIFTS is crucial in elucidating gas sensing mechanisms by providing

**Table 5**Single atom-engineered VOC detection by various sensor techniques, detection limit, and working temperature.

S. No.	Type of sensor	VOC analyte	Support material	Detection limit (ppb)	Working temperature (°C)	Ref
1.	Metal oxide semiconductor (MOS) gas sensor	Formaldehyde	Au/In <sub>2</sub> O <sub>3</sub>	1.42	100	[105]
2.	Metal oxide semiconductor (MOS) gas sensor	Methanol	Ag-LaFeO <sub>3</sub> @ZnO-Pt	3.27	86	[148]
3.	Metal oxide semiconductor (MOS) gas sensor	Acetone	SA-Pt/WO <sub>3-x</sub>	4.3	350	[108]
4.	Semiconductor-based gas sensor	Formaldehyde	Pt on MOF-derived In <sub>2</sub> O <sub>3</sub>	8.4	-	[73]
5.	Metal oxide semiconductor (MOS) gas sensor	Xylene	3DOM PtCo-WO <sub>3</sub>	1.08	250	[150]
6.	Metal oxide semiconductor (MOS) gas sensor	Xylene	3DOM PtNi-WO <sub>3</sub>	1.69	250	[150]
7.	Chemiresistive gas sensor	Toluene	Cu SA/WO <sub>2.72</sub>	10	160	[146]
8.	Metal oxide semiconductor (MOS) gas sensor	Triethylamine	SA-Pt/WO <sub>3</sub>	0.18	240	[149]
9.	Field-effect transistor (FET) gas sensor	Triethylamine	Pt SA-Ti <sub>3</sub> C <sub>2</sub> T <sub>x</sub>	14	RT	[70]
10.	Chemiresistive gas sensor	Ethanol	Pt SAs-MoS <sub>2</sub>	5 (ppm)	RT	[160]
11.	Metal oxide semiconductor (MOS) gas sensor	Formaldehyde	SA Rh-sensitized SnO <sub>2</sub>	55	250	[74]
12.	Metal oxide semiconductor (MOS) gas sensor	Triethylamine	Pt Single atom loaded ZnO	17.8	200	[144]
13.	Chemiresistive gas sensor	Formaldehyde	PtSA-SnO <sub>2</sub>	6	175	[151]
14	Microelectromechanical system (MEMS)	3-hydroxy-2-	SnO <sub>2</sub> /Au-SA	10	50	[158]
	sensor	butanone				

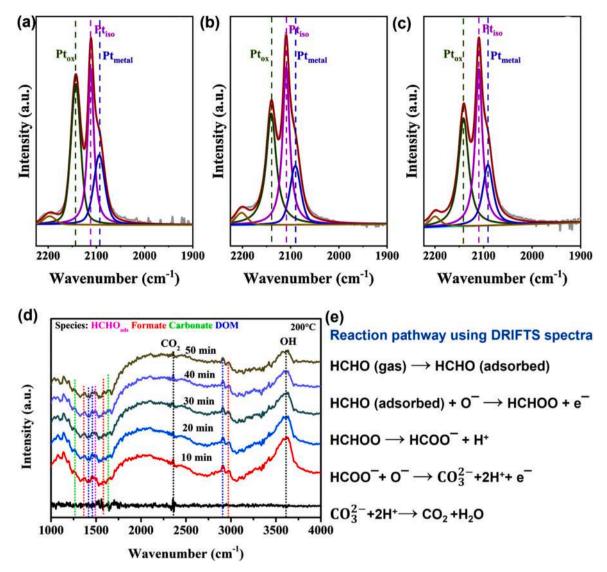


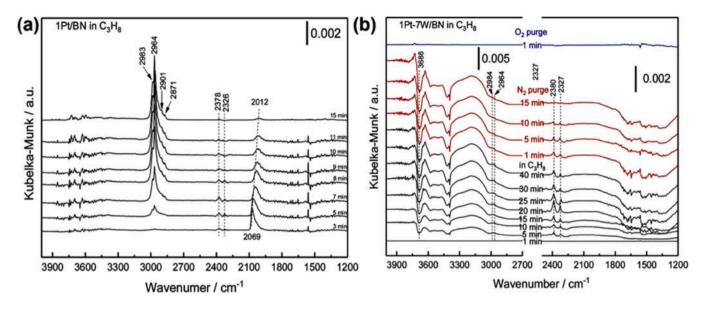
Fig. 11. CO-DRIFTS spectra of (a)  $Pt@WO_3$ , (b)  $PtCo@WO_3$ , and (c)  $PtNi@WO_3$ , highlighting surface interactions Reprinted with permission from Ref. [150]. (d) In situ, DRIFTS spectra of  $Pt@In_2O_3$  exposed to 100 ppm HCHO at 200 °C, capturing the formation and evolution of key reaction intermediates. (e) Representative oxidation mechanism of HCHO proposed based on DRIFTS analysis. Reprinted with permission from Ref. [73].

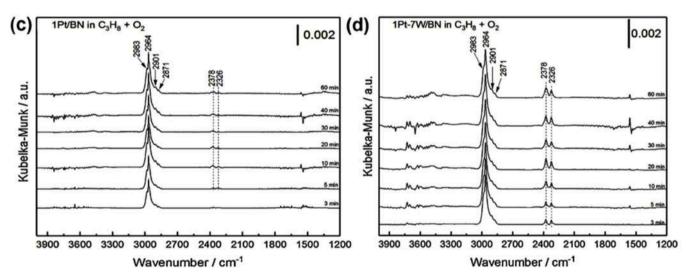
real-time insights into surface interactions in SA catalysts. DRIFTS enhances the understanding of SA-based gas-sensing applications by identifying key surface intermediates, tracking reaction kinetics, and distinguishing atomic configurations. The real-time monitoring capabilities of DRIFTS have been effectively demonstrated in the study of Pt-In<sub>2</sub>O<sub>3</sub> for HCHO detection [73]. As shown in Fig. 11d, exposure to 100 ppm HCHO at 200°C yields time-resolved in situ DRIFTS spectra with distinct vibrational peaks corresponding to key oxidation intermediates. Adsorbed HCHO (1495 cm<sup>-1</sup>) undergoes transformation into formate (1356, 1584, and 2968 cm<sup>-1</sup>), carbonate (1258 and  $1645~\mathrm{cm}^{-1}$ ), and dioxymethylene species (1410, 1474, and 2872  $\mathrm{cm}^{-1}$ ), alongside hydroxyl species (3620 cm<sup>-1</sup>) and CO<sub>2</sub> (2359 cm<sup>-1</sup>). These spectral changes confirm a sequential oxidation pathway ultimately leads to the formation of CO<sub>2</sub> and H<sub>2</sub>O. The corresponding resistance variations observed in gas-sensing measurements align with these reaction steps, providing mechanistic insights into the role of surface-adsorbed oxygen in catalytic oxidation.

The reaction pathway, further supported by the proposed equations (Fig. 11e), deepens the fundamental understanding of the VOC oxidation mechanism in SA-based sensors. In situ DRIFTS conducted under oxygen-deficient and oxygen-rich conditions provides direct evidence of

the distinct roles played by adsorbed oxygen species and lattice oxygen, offering a comprehensive understanding of their dynamic contributions to the catalytic process. For example, CO-DRIFTS studies during propane oxidation identify CO as a critical intermediate, underscoring the mechanistic involvement of oxygen species [144].

As shown in Fig. 12a, shifts in CO adsorption bands (2069–2012 cm) confirm the active participation of surface oxygen, while the emergence of hydroxyl-related bands (3688–3392 cm<sup>-1</sup>) on WO<sub>3</sub> highlights their direct involvement in VOC oxidation. In oxygen-deficient conditions (Fig. 12a, b), CO forms initially on metallic Pt sites, indicating partial oxidation of propane. However, in Pt-WO<sub>3</sub>/BN catalysts, the gradual depletion of hydroxyl bands (3688–3392 cm<sup>-1</sup>) correlates with propane activation and subsequent oxidation to CO<sub>2</sub> (2326–2378 cm<sup>-1</sup>), emphasizing the essential role of hydroxyl species in the reaction pathway. Temporal variations of CO and CO<sub>2</sub> band intensities reflect the dynamic consumption and regeneration of reactive oxygen species, illustrating the role of WO3 in sustaining catalytic oxidation through oxygen renewal. Upon reintroduction of oxygen (Fig. 12c, d), the rapid structural recovery of WO3 validates its contribution to catalytic stability and efficiency. These findings highlight that both oxygen and hydroxyl groups on WO3 are critical to the oxidation mechanism,





**Fig. 12.** In-situ DRIFT spectra for propane oxidation over (a, b) Pt/BN and Pt-WO<sub>3</sub>/BN catalysts without oxygen, showing CO formation as an intermediate and hydroxyl group consumption on WO<sub>3</sub>. (c, d) Spectra with oxygen present, highlighting increased CO<sub>2</sub> production and reversible structural restoration of Pt-WO<sub>3</sub>/BN Reprinted with permission from Ref. [144].

particularly at the Pt-WO<sub>3</sub> interface, where they facilitate rate-determining steps and enhance overall catalytic activity. In situ DRIFTS thus serves as a vital tool for probing the electronic and structural characteristics of SACs. Correlating spectroscopic features with gas-sensing performance enables the rational design of high-performance VOC sensors. When combined with complementary techniques such as XPS and HAADF-STEM, DRIFTS enhances the robustness of experimental insight, offering a powerful framework for advancing SA-catalyst research.

# 4.1.2. Significance of in situ Raman Spectroscope in VOC Spectroscopy

Raman spectroscopy is a powerful technique for probing the structural and electronic properties of SACs. It offers valuable insights into defect states, chemical coordination, and the local environment of active sites. Specifically, Raman analysis can sensitively detect oxygen vacancies, strain effects, and electronic interactions with adsorbates, making it highly relevant for elucidating VOC sensing mechanisms. While *ex situ* Raman spectroscopy has been widely applied to characterize SACs in VOC-related studies, *in situ* and operando implementations remain relatively less explored. *In situ* Raman spectroscopy enables

real-time monitoring of catalyst evolution under working conditions, offering a deeper understanding of reaction intermediates, structural transformations, and active site dynamics. This technique has been extensively utilized in electrochemical energy systems, including HER, ORR, OER, and batteries, where it has revealed crucial phenomena such as active site restructuring and degradation pathways. However, its potential in VOC sensing has yet to be fully realized.

For example, in HER and OER studies, *in situ* Raman spectroscopy has been employed to track the formation of hydroxide intermediates and their interactions with active sites. Similarly, in ORR investigations, the evolution of oxygen-containing species during the reaction process. Applying this methodology to VOC sensing could provide analogous insights by enabling the detection of VOC adsorption, intermediate species formation, and real-time transformation under sensing conditions.

Future work should aim to adapt *in situ* Raman techniques for VOC sensing applications to bridge this critical knowledge gap. Integration with complementary characterization tools such as XPS, XANES, and EXAFS can provide a more holistic mechanistic understanding, facilitating the rational design of high-performance SAC-based VOC sensors.

Furthermore, combining *operando* Raman spectroscopy with electrical resistance measurements could establish direct correlations between spectral changes and sensing behavior, thereby accelerating progress in SAC-enabled gas sensing technologies.

# 4.2. Computational approaches: DFT and Machine learning for SA VOC sensors

DFT plays a central role in understanding the electronic and geometric structures of SACs. It provides detailed information on adsorption energies, charge transfer mechanisms, and reaction pathways. In the context of VOC sensing, DFT enables the analysis of adsorption behavior, reaction mechanisms, and defect chemistry by calculating charge transfer dynamics, energy barriers, transition states, and formation energies. DFT has been extensively applied to 2D materials such as TMDs, including WS<sub>2</sub>, MoS<sub>2</sub>, SnS<sub>2</sub>, and CrX<sub>2</sub>, as well as dual-atom doped graphdiyne and 2D C<sub>2</sub>N, owing to their tunable electronic structures, high surface areas, and strong metal-support interactions [161–165]. These properties make them highly suitable for electrocatalysis, energy storage, and gas sensing applications.

To evaluate the thermodynamic stability of SACs on supporting materials, the binding energy  $(E_b)$  is calculated as [166],

$$E_b = E(SA@Substrate) - E(Substrate) - E(SA)$$
 (1)

where E(SA@Substrate) is the total energy of the SA-loaded system, E (Substrate) is the energy of pristine support, and E(SA) is the energy of the isolated SA.

A recent DFT study investigated the adsorption of eleven VOCs on pristine  $WS_2$ , sulfur-vacancy-induced  $WS_2$  ( $V_S$ - $WS_2$ ), and metal-doped  $WS_2$  (M@WS<sub>2</sub>, where Co, Fe, Nb, and Ni). After geometry optimization, the VOC adsorption energy on  $WS_2$  was evaluated using an equation.

demonstrating strong chemisorption, registering adsorption energies of -2.09 eV, -2.08 eV, and -1.89 eV, respectively. Bond lengths in the range of 1.96--2.35 Å further supported strong metal-VOC interactions. For optimal gas sensing, the adsorption energy should ideally lie between strong physisorption and weak chemisorption, with  $E_{ads}$  values in the range of -1.20 to -2.00 eV [167,168]. While pristine WS $_2$  and  $V_S\text{--WS}_2$  fell outside this optimal range, metal doping, particularly with Nb-enabled favorable VOC adsorption, positioned Nb@WS $_2$  as a promising candidate for next-generation VOC sensors. Overall, DFT offers critical insights into the intrinsic properties of the SAC system, guiding experimental efforts and mechanistic understanding, as discussed in the previous section.

In parallel, ML has emerged as a prevailing tool to accelerate catalyst discovery and establish structure-activity relationships in SACs [169]. By leveraging extensive experimental and theoretical datasets, ML enhances predictive accuracy and computational efficiency, thereby accelerating the discovery of novel materials and improving the sensing performance of previously untested VOCs. ML-driven approaches facilitate the rapid screening of SAs materials by predicting key parameters such as adsorption energies, activation barriers, and charge transfer properties, enabling the identification of promising candidates with high catalytic activity. Moreover, ML models assist in recognizing key descriptors that govern SAC performance, such as coordination environments, electronic structures, and defect densities, thereby guiding the rational design of optimized catalysts [170]. Beyond accelerating the discovery process, ML-based models and neural networks effectively reduce the computational cost of DFT simulations while preserving accuracy, making large-scale screening of SAC materials feasible. In the context of VOC sensing, ML models have also played a crucial role in improving detection accuracy, selectivity, and response time [171]. Commonly employed models such as Principal Component Analysis (PCA), Support Vector Machines (SVM), Random Forest (RF), k-Nearest Neighbors (k-NN), Artificial Neural Networks (ANN), and Convolutional Neural Networks (CNN) are used to extract features, classify sensor re-

$$E_{ads} = E(WS_2withV_s - WS_2 \quad withM@WS_2@VOC) - E(WS_2withV_s - WS_2 \quad withM@WS_2) - E(VOCs)$$
 (2)

where  $E(WS_2 \text{ with } V_S-WS_2 \text{ with } M@WS_2@VOCs)$  is the total energy of the VOC-adsorbed system,  $E(WS_2 \text{ with } V_S-WS_2 \text{ with } M@WS_2)$  is the total energy of the substrate, and E(VOCs) is the energy of the isolated VOC molecule.

Adsorption energy trends revealed that octane exhibited the strongest interaction at -0.78~eV, followed by heptane at -0.72~eV, toluene and benzaldehyde (-0.61 eV each), while methylamine, 2-butanone, and dimethyl disulfide demonstrated weaker adsorption with values of  $-0.26~eV,\ -0.41~eV,\$ and  $-0.45~eV,\$ respectively (Fig. 13a). Partial density of states (PDOS) analysis revealed minor shifts due to contributions from O(p), N(p), C(p), and S(p) orbitals of the VOC molecules. The Fermi level shift towards the conduction band minimum indicated n-type semiconductor behavior (Fig. 13b). However, Bader charge analysis and electron localization function plots revealed negligible charge transfer and weak van der Waals interactions in pristine WS $_2$  (Fig. 13c).

To enhance VOC adsorption, a single sulfur vacancy ( $V_S$ -WS<sub>2</sub>) was introduced, resulting in a formation energy of -6.19 eV. However, the corresponding adsorption energies ( $E_{ads}$ ) showed marginal improvements, suggesting that defect engineering alone may not significantly enhance sensing performance. In contrast, transition metal doping led to substantial improvements; binding energies for Co, Fe, Ni, and Nb dopants were calculated as -4.72 eV, -4.06 eV, -4.88 eV, and -8.47 eV, respectively. Among these, Nb-doped WS<sub>2</sub> exhibited the highest affinity for VOCs, with 2-butanone, butanol, and hexanal

sponses, and identify VOC species with improved precision [172].

To illustrate the potential of ML in VOC sensing, advanced ML models like RF, CNNs, and SVMs have demonstrated exceptional performance in gas detection, offering insights applicable to SAC-based systems. RF, an ensemble-based model, aggregates decisions from multiple decision trees to achieve robust classification, even under sensor drift or low signal-to-noise conditions. In a study using SnO<sub>2</sub>-based sensor arrays, RF achieved 99 % accuracy in distinguishing structurally similar VOCs, with feature importance metrics (e.g., adsorption energy) highlighting key descriptors relevant to SAC design [173,174].

$$Importance(f_i) = \sum_{t \in T(f_i)} \frac{N_t}{N} \quad \Delta i_t$$

where  $T(f_i)$  is the set of nodes where  $f_i$  is used,  $N_t$  is the number of samples at node t, and  $\Delta i_t$  represents the impurity reduction. Similarly, CNNs excel at extracting spatiotemporal features from raw sensor signals, reducing reliance on manual feature engineering. For instance, a CNN framework integrating time-variant illumination in a  $\mu$ LED-based sensor achieved ~97 % accuracy in identifying methanol, ethanol, acetone, and  $NO_2$  at sub-ppm levels, while another CNN model using grayscale-transformed resistance-time matrices classified acetone and formaldehyde with 100 % accuracy [175,176]. The CNN convolution operation is defined as [177]:

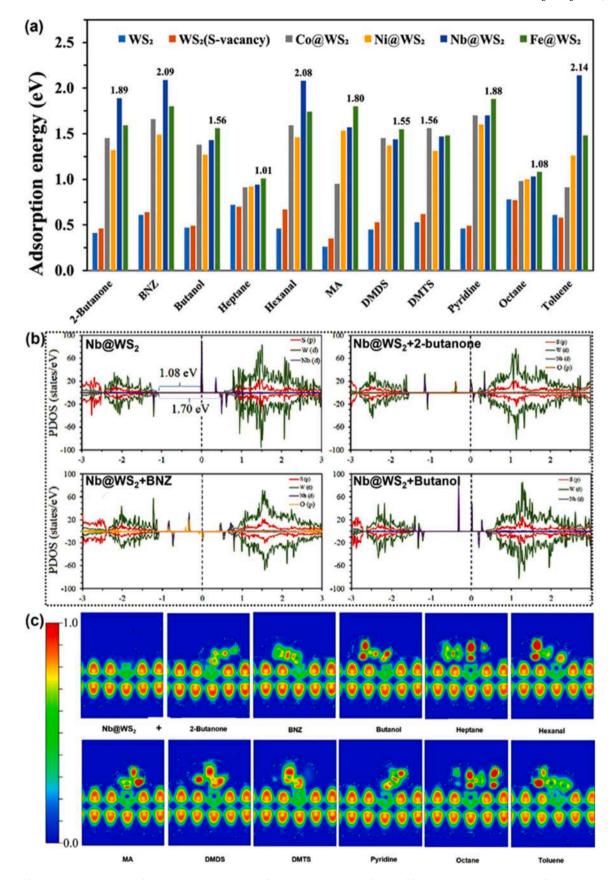


Fig. 13. (a) Adsorption energies ( $E_{ads}$ ) of VOCs on WS<sub>2</sub>, VS-WS<sub>2</sub>, and M@WS<sub>2</sub> (M=Co, Fe, Nb, Ni). (b) DOS variations upon VOC adsorption. (c) ELF plots of Nb@WS<sub>2</sub> and its interactions with various VOCs, illustrating electron distribution (0: absent, ~0.5: uniform, 1: localized). Reprinted with permission from Ref. [161].

$$Output(i, \quad j) = \sum_{m=0}^{M-1} \sum_{n=0}^{N-1} Input(i+m, j+n) \quad .K(m, n)$$

SVMs, effective in low-data regimes, achieved up to 100% accuracy in hybrid RF-SVM frameworks, using kernel-based decision functions [178,179].

$$f(x) = sign(\sum_{i=1}^{n} \alpha_i y_i K(x_i, x) + b)$$

These high accuracies (97-100 %) in non-SAC sensing materials suggest that RF, CNNs, and SVMs, when combined with DFT-derived SAC descriptors (e.g., adsorption energy, charge transfer), could significantly enhance the selectivity and sensitivity of SAC-based VOC sensors. Table 6 summarizes a range of ML models applied to conventional VOC sensors, highlighting their potential and outcomes, which serve as a foundation for advancing SAC-based sensing applications [173, 175, 176, 180-185]. Drawing on the ML approaches summarized in Table 6 [186], the integration of such models into SAC-based VOC sensor systems can follow a systematic workflow to ensure optimal performance, as shown in Fig. 14a. The process begins with data acquisition, where sensor response data is collected under controlled conditions to capture baseline signals and VOC-specific responses while monitoring environmental parameters like temperature and humidity to reduce noise. In the preprocessing stage, techniques such as data normalization, scaling, and outlier detection are applied to enhance data consistency and improve model reliability. Subsequently, feature extraction and selection methods, such as PCA or Recursive Feature Elimination (RFE), are utilized to identify critical features while eliminating redundant information. The refined dataset is then employed for model training and validation, where supervised algorithms are trained with labeled data and assessed using cross-validation methods to prevent overfitting and ensure model robustness. The trained model is then applied for prediction and classification, where it classifies unknown VOC samples based on sensor response patterns, improving real-time detection in practical applications. Following this, performance evaluation using metrics such as accuracy, precision, recall, and F1-score (combined precision and recall) ensures comprehensive model assessment, while confusion matrix analysis aids in refining model predictions. Lastly, the model is integrated into the sensor system for deployment and monitoring, enabling real-time VOC detection with periodic recalibration to maintain performance stability. Fig. 14b presents a representative scheme highlighting the role of ML in gas sensor detection and analysis of unknown gas inputs [187]. This integrated ML framework strengthens VOC sensor systems, ensuring precise detection, improved stability, and enhanced data interpretation, advancing their role in environmental monitoring, industrial safety, and healthcare applications.

In the discovery of new materials, the integration of DFT-calculated parameters with ML models has

shown significant potential in improving VOC sensor performance by enhancing predictive accuracy and identifying key material features. In a representative study, Hu and co-workers employed a data-enhanced network strategy to optimize gas-sensitive materials [188]. The workflow began with DFT calculations to extract critical parameters such as adsorption energy, bond distances (d), and structural distortions ( $\Delta$ d), which were used as input features for model training (Fig. 15a). Eight ML classifiers, including convolutional neural network (CNN), multilayer perceptron (MLP), random forest (RF), logistic regression (Logist), decision tree (Tree), support vector machine (SVM), bagging classifier (Bagging), and voting classifier (Voting), were evaluated to determine the most effective predictive model (Fig. 15b). Fig. 15c shows that the voting classifier achieved the highest accuracy (85.71 %), surpassing bagging (57.14 %) and other models (71.43 %). The receiver operating characteristic (ROC) curve analysis further validated the

robustness of the voting classifier, with its area under the curve (AUC) values closely approaching 1, signifying superior performance. Further, feature analysis identified adsorption energy, a key descriptor derived from DFT data, as one of the most influential features in predicting VOC response. Pearson correlation analysis and feature importance ranking reinforced that Eads, along with selected structural parameters (d and  $\Delta$ d), played a pivotal role in enhancing model performance (Fig. 15 d, e). The confusion matrix shown in Fig. 15f further confirms the voting classifier's high prediction accuracy and robustness against misclassification. This approach highlights the synergistic combination of DFT data and ML algorithms, demonstrating how electronic structure insights can guide the rational design of VOC sensors with improved accuracy, selectivity, and stability. By leveraging datadriven strategies, VOC sensor development can be accelerated, offering innovative solutions for environmental monitoring and industrial safety.

While ML models have been extensively explored for VOC sensors [189], their application in SAC-based VOC sensors remains relatively limited. Leveraging DFT-derived SAC parameters, such as adsorption energies, charge transfer characteristics, and geometric descriptors, can significantly enhance the predictive capabilities of ML models in this

**Table 6**Representative ML model in VOC sensing.

S. No	ML Model	Category	Potentials	Target VOCs	Case Sensor Systems	Ref
1	Random Forest	Supervised (Ensemble)	Robust to noise, ranks feature importance	Formaldehyde, methanol, 2-propanol, toluene, acetone, benzene, ethanol	SnO <sub>2</sub> -based and noble-metal- decorated sensor arrays	[175,176]
2	Support Vector Machines	Supervised (Kernel-based)	Effective for small dataset, high precision	Benzene, toluene, acetone, ethanol	Pd-glass, zeolite–oxide systems, Metal oxides	[174,175, 180,181]
3	Convolution Neural Network	Deep Learning	Learns from raw signals, high dimensionality support	Ethanol, acetone, methanol, formaldehyde, benzene	$\mu$ LED dynamic sensors, In <sub>2</sub> O <sub>3</sub> -based sensors	[182,183]
4	Artificial Neural Network	Supervised (Neural)	Capture non-linear relationships	Ethanol, Formaldehyde, benzene, toluene	Noble-metal In <sub>2</sub> O <sub>3</sub> arrays, Cr and Pd doped In <sub>2</sub> O <sub>3</sub>	[183]
5	Multilayer Perception	Supervised, Feedforward	Flexible architecture for small to mid-size datasets	Methanol, toluene, formaldehyde, 2- propanol	SnO <sub>2</sub> and hybrid oxide sensors	[175]
6	Linear Regression	Supervised (Linear)	Interpretable, suitable for concentration estimation	Acetone, benzene, ethanol, toluene, formaldehyde, isobutanol	Metal oxide associated MEMS-based VOC sensor array	[184,185]
7	Principal Component Analysis	Unsupervised Feature Reduction	Reduces data dimensionality; enhances model performance	Acetone, ethanol, benzene, toluene formaldehyde, methanol	ZnO, In <sub>2</sub> O <sub>3</sub> -based sensor arrays	[180,183, 186]
8	K-Nearest Neighbor (k-NN)	Supervised, Distance-based	Easy to implement; no model training required	Ethanol, methanol, acetone	rGO/CuCoOx sensor system, metal oxides	[186] [190]
9	Deep Neural Network	Deep Learning	Learns complex patterns; scalable to multi-gas and mixture data	Ethanol, formaldehyde, benzene, toluene	Cr/Pd-doped In <sub>2</sub> O <sub>3</sub> arrays	[183]

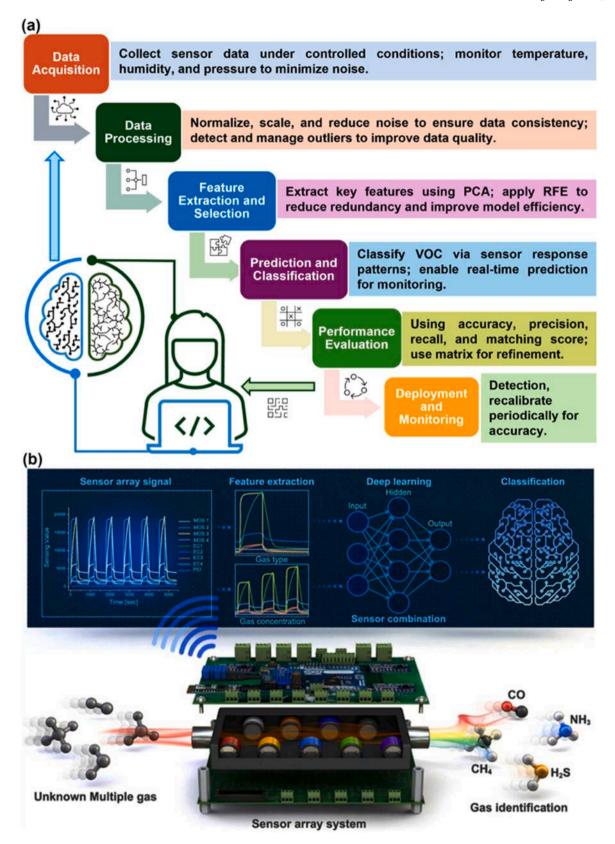


Fig. 14. (a) Machine learning framework for VOC detection and sensor material design. (b) Example of an integrated sensor system demonstrating the detection of unknown gases Reprinted with permission from Ref. [186].

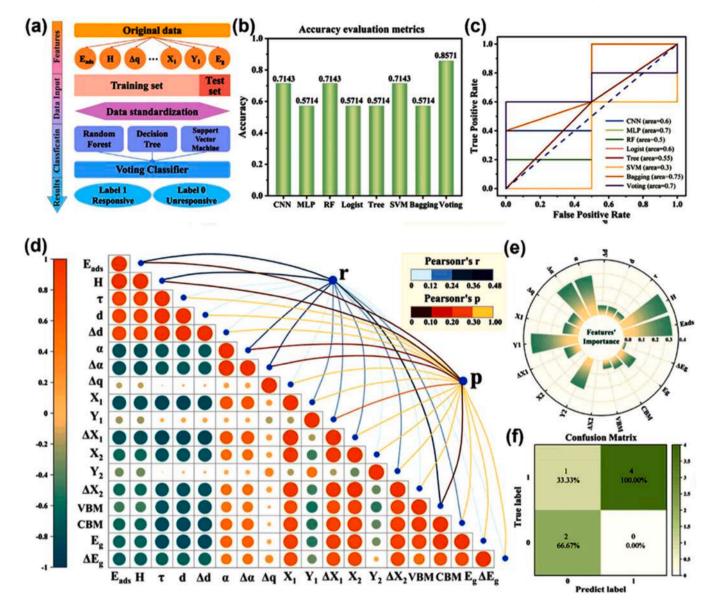


Fig. 15. Overview of the ML-based analysis framework for VOC sensing; (a) Workflow depicting data acquisition, preprocessing, model training, and evaluation. (b) Classification model accuracy comparison for VOC detection. (c) Receiver operating characteristic (ROC) curve highlighting model performance. (d) Pearson correlation analysis illustrating feature relevance to VOC classification. (e) Statistical significance of 18 selected features influencing sensor response. (f) Confusion matrix demonstrating model prediction outcomes for the test dataset Reprinted with permission from Ref. [188].

context. For instance, Liu et al. utilized ML to evaluate metal-2D SACs and predict their response intensity in SAC-based chemical sensors [158].

The gradient boosting regression (GBR) model exhibited superior predictive performance when incorporating four key descriptors; charge transfer ( $\delta_{CT}$ ), metal charge ( $q_M$ ), metal-oxygen distance ( $D_{M\text{-}O}$ ), and VOC type. Unlike conventional ML models that primarily predict binding energies (Fig. 16a), this approach introduced VOC classification (e. g., acetone, ethanol, and RH labeled as 1, 2, and 3, respectively) to refine response intensity predictions. As depicted in Fig. 16b, the integration of these four descriptors enabled accurate prediction of response intensity, achieving a Pearson correlation coefficient of approximately 0.79, thereby validating the model's robustness.

The integration of DFT with ML-driven methodologies offers a synergistic pathway for tailoring SACs with enhanced properties for VOC sensing. Future advancements in self-learning models,

capable of iteratively refining theoretical predictions based on experimental feedback, hold the potential to revolutionize the field,

paving the way for more intelligent and efficient catalyst design strategies.

# 5. Current state of the art, limitations and future prospects

Although significant progress has been made in the development of SAES for VOC detection, further advancements are required to move beyond proof-of-concept studies and achieve reliable, real-world applications. This section outlines key research gaps, grounded in current successes and limitations, and proposes well justified research directions for future studies (Table 7).

# Achievement: SAES exhibits ultra-high sensitivity and atom-efficient design.

Problem: Many VOCs possess similar molecular weights, chemical structures, and binding affinities, making it extremely difficult to differentiate between analytes like methanol, ethanol, benzene, and toluene. In real-world environments, temperature and humidity

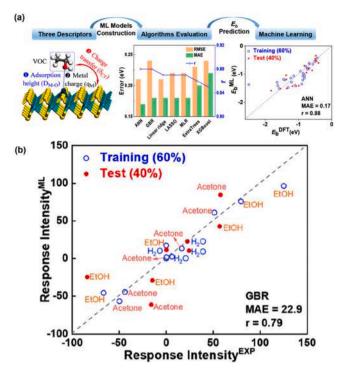


Fig. 16. (a) Schematic representation of ML-based prediction of  $E_{ads}$  using three key descriptors ( $\delta_{CT}$ ,  $q_{M}$ , and  $D_{M\cdot O}$ ), with performance assessed via RMSE, MAE, and r, highlighting ANN as the most effective model. (b) Comparison of ML-predicted and experimentally observed values using four descriptors ( $\delta_{CT}$ ,  $q_{M}$ ,  $D_{M\cdot O}$ , and VOC). Reprinted with permission from Ref. [158].

fluctuations further impair performance.

Future Research: Develop selectivity tuned SAES by engineering the coordination environment of the single atoms (e.g., via N, S, or P heteroatom anchoring) to introduce analyte specific recognition. Tune site specific functionalization (e.g., -OH, -COOH and -NH $_2$  groups) on the sensor surface to create chemical fingerprints for VOCs. Construct multichannel sensor arrays (e- nose systems) with different SA sites, coupled with machine learning algorithms for cross-analyte recognition under variable humidity and temperature.

2. Achievement: Atomic dispersion ensures maximum atom utilization and catalytic performance.

Problem: The high surface energy of single atoms leads to agglomeration or migration, particularly under operational conditions or long-term storage, resulting in loss of sensitivity.

Future Research: Apply defect-engineered supports (e.g., graphene vacancies, oxygen-rich MXenes, or doped carbon frameworks) to strongly anchor single atoms and prevent mobility. Investigate dual-atom or cluster-based strategies, where synergistic interactions between adjacent atoms prevent aggregation while enhancing sensing performance. Develop real-time operando STEM imaging and thermal stability studies to understand atom dynamics and sensor degradation over time.

Achievement: SAES shows rapid response and recovery times for real-time sensing.

Problem: Signal fluctuations and poor reproducibility are observed in real-world scenarios due to interference from environmental factors and the dynamic interaction of VOCs with atomic sites.

Future Research: Integrate dynamic compensation mechanisms, such as humidity tolerant layers or self-calibration feedback loops, in sensor devices. Embed nanoelectronic transducers (e.g., FET- based architectures) for enhanced signal-to-noise ratios and stable outputs. Use adaptive learning systems that train the sensor response over time for drift correction and stable performance in field settings.

**Table 7**Summary of key challenges and potential solutions in SAES for VOC detection

Challenges	Impact	Possible solutions
Agglomeration of single atoms	Loss of activity; cluster formation	Use of strong anchoring sites (N, O, S); defect engineering
		MOF-derived and carbon supports
Low metal loading	Insufficient active sites;	Optimized SA dispersion;
limits	low signal strength	dual-atom or cluster engineering
Uncontrolled	Inconsistent sensor	Advanced synthesis methods
coordination	performance	like ALD, wet-chemistry
environment		control, and coordination- tuned precursors
Thermal and chemical	Degradation during	Design of thermally robust
instability	sensing cycles	supports; hydrophobic
Overlanning hinding	Poor celectivity	coatings Functional group
Overlapping binding affinities among	Poor selectivity	engineering; heteroatom
VOCs		doping; sensor arrays with A classifiers
Environmental	False signals or signal	Surface passivation;
interference (e.g.,	drift	hydrophobic coatings;
humidity, CO <sub>2</sub> )		machine learning correction
,, 552)		algorithms
Weak signal intensity	Low sensitivity	Signal amplification
- 0		techniques (e.g., transducers
		hybrid sensing platforms)
rreversible VOC	Loss of reusability and	Reversible binding designs;
binding	sensitivity	optimal binding energy
-	•	tuning (via DFT)
Complex and	Poor reproducibility	Green, scalable protocols like
inconsistent		electrospinning, spray
synthesis		pyrolysis, ionic liquids
Limited real-time/	Incomplete	Adoption of operando XAS,
operando	understanding of	XPS, DRIFTS, Raman, EPR
mechanistic data	working principles	
Lack of advanced	Difficulty confirming SA	STEM-EELS, HAADF-STEM,
characterization	dispersion or VOC	EXAFS, in-situ spectroscopy
	interaction	
Idealized DFT models	Theoretical predictions	Combine DFT with solvent
	deviate from	effects, machine learning, or
	experiments	experimental data
Trial-and-error	Low efficiency in sensor	ML-guided material
material screening	design	selection; high-throughput
Chart amountianal	Limited real-world	DFT screening
Short operational lifespan		Durability testing; encapsulation;
шезрап	utility	environmental shielding
incompatibility with	Hinders development of	Use of SA-hybrid materials or
flexible substrates	wearable/flexible	polymer or 2D conductive
HEADIC SUDSTIALES	sensors	platforms
Poor signal	Inaccurate detection;	Device standardization and
reproducibility	unreliable calibration	real-time calibration
·r · · · · · · · · · · · · · · · · · ·		mechanisms
Difficult film	Batch-to-batch variation;	Blade coating, inkjet
uniformity during	noisy signals	printing, or self-assembled
deposition	, U	monolayer techniques
Low scalability of	Barrier to industrial	Scalable green routes:
synthesis	translation	solvothermal, self-assembly,
		and spray deposition
	Expensive and limited	Explore earth-abundant
High reliance on noble		
High reliance on noble metals	resource availability	metals (Fe, Co, Ni, Mn, Cu, Al)
metals	•	Al)
metals	cost outweighs performance gains	Al) Optimize sensor performance
metals  Poor technoeconomic	Cost outweighs	Al) Optimize sensor performance per unit metal; replace noble
metals Poor technoeconomic viability	Cost outweighs	Al) Optimize sensor performance per unit metal; replace noble metals with cost-effective
metals Poor technoeconomic viability	Cost outweighs performance gains	Al) Optimize sensor performance per unit metal; replace noble metals with cost-effective alternatives
metals  Poor technoeconomic viability  Limited validation	Cost outweighs performance gains	Al) Optimize sensor performance per unit metal; replace noble metals with cost-effective alternatives Conduct testing in real air
metals  Poor technoeconomic viability  Limited validation under real sample conditions	Cost outweighs performance gains  Lab results may not translate to field	Al) Optimize sensor performance per unit metal; replace noble metals with cost-effective alternatives Conduct testing in real air samples from industry,
metals  Poor technoeconomic viability  Limited validation under real sample conditions	Cost outweighs performance gains  Lab results may not translate to field conditions	Al) Optimize sensor performance per unit metal; replace noble metals with cost-effective alternatives Conduct testing in real air samples from industry, hospitals, and cities
metals  Poor technoeconomic viability  Limited validation under real sample conditions Lack of standard	Cost outweighs performance gains  Lab results may not translate to field conditions Hard to compare SAES	Al) Optimize sensor performance per unit metal; replace noble metals with cost-effective alternatives Conduct testing in real air samples from industry, hospitals, and cities Develop standard test
metals  Poor technoeconomic viability  Limited validation under real sample conditions Lack of standard benchmarking protocols	Cost outweighs performance gains  Lab results may not translate to field conditions Hard to compare SAES	Al) Optimize sensor performance per unit metal; replace noble metals with cost-effective alternatives Conduct testing in real air samples from industry, hospitals, and cities Develop standard test protocols (gas exposure,
Poor technoeconomic viability  Limited validation under real sample conditions  Lack of standard benchmarking	Cost outweighs performance gains  Lab results may not translate to field conditions Hard to compare SAES across studies	Al) Optimize sensor performance per unit metal; replace noble metals with cost-effective alternatives Conduct testing in real air samples from industry, hospitals, and cities Develop standard test protocols (gas exposure, humidity control, stability)

(continued on next page)

Table 7 (continued)

Challenges	Impact	Possible solutions
Insufficient integration into devices	Standalone materials, not usable devices	Integrate SAES into complete sensor platforms (chips, wireless modules, IoT systems)
Environmental and disposal issues	Toxicity and sustainability concerns	Use of eco-friendly synthesis, biodegradable supports, and recycling strategies

Achievement: Proof of concept SAES developed in laboratory settings.

Problem: The scalability and process reproducibility of these sensors remain unproven for large- scale industrial production.

Future Research: Develop green, low-temperature, and solution-processable synthesis routes such as UV-assisted photochemical reduction, ionic-liquid-mediated synthesis, electrospinning, or ultrasonic spray pyrolysis. Standardize scalable deposition techniques (e.g., inkjet printing, roll to roll coating) for sensor fabrication on flexible substrates. Design pilot-scale demonstration modules and field-testing prototypes to validate stability and reproducibility over months of operation.

Achievement: Initial mechanistic insights from DFT and basic spectroscopy.

Problem: The fundamental sensing mechanism, including electron transfer, adsorption geometry, and reaction intermediates, remains poorly understood at the atomic scale.

Future Research: Employ advanced in situ and operando spectroscopies (e.g., XPS, XAS, DRIFTS, EPR, Raman, and STEM) during VOC exposure to capture real-time changes in SA oxidation state and bonding configuration. Correlate experimental data with DFT calculated adsorption energies, charge distributions, and orbital overlaps for various VOCs on specific SA sites. Explore transient techniques (e.g., pulsed voltammetry-based sensing or TPD-MS coupling) to monitor fast electron transfer and desorption kinetics.

Achievement: DFT modeling has begun to assist material selection and screening.

Problem: The trial and error approach remains dominant; DFT alone cannot easily predict complex, real-world behavior involving competing analytes and interference.

Future Research: Combine machine learning (ML) with DFT data-bases to predict optimal metal support ligand configurations tailored for VOC sensing. Apply inverse design models to propose novel SAES systems with target performance metrics (e.g., selectivity for benzene in the presence of toluene). Create open-source knowledge platforms integrating theoretical predictions with experimental feedback to guide community wide sensor optimization.

#### 5.1. Conclusions

Single-atom engineered sensors (SAES) represent a transformative advancement in the field of VOC detection due to their unparalleled atom efficiency, tunable electronic structures, and precise molecular recognition capabilities. Through the engineering of isolated atomic sites and their surrounding coordination environments, SAES provides superior sensitivity and selectivity, enabling the detection of trace level VOCs with rapid response and minimal cross interference. These attributes make them particularly suited for applications ranging from environmental monitoring to food safety and biomedical diagnostics. However, realizing the full potential of SAES demands overcoming persistent challenges related to their synthesis, long term stability, and large-scale fabrication. Preventing atom agglomeration, ensuring robust anchoring, and addressing environmental interferences remain critical hurdles. Furthermore, an incomplete mechanistic understanding of SAES VOC interactions continues to limit rational sensor design. In this

regard, in situ and operando spectroscopies, along with density functional theory (DFT) calculations and machine learning algorithms, are essential tools for decoding the atomistic pathways that govern sensor behavior and performance. Looking forward, interdisciplinary efforts that integrate materials chemistry, surface science, and data driven modeling are crucial to optimize SAES for real world deployment. The use of earth abundant metals, advanced defect engineering, and dual or multi atom configurations will broaden the design space and functional applicability of these sensors. Furthermore, establishing industrially relevant testing protocols will be vital for scaling SAES technologies from laboratory prototypes to commercial sensor platforms. In essence, while SAES are still in their developmental stage, their promise is undeniable. With continued innovations in material design and mechanistic insight, SAES is poised to redefine how we monitor and respond to volatile organic pollutants in increasingly complex environments.

#### CRediT authorship contribution statement

Sowjanya Vallem: Writing review & editing, Writing original draft, Visualization, Conceptualization. Malayil Gopalan Sibi: Writing review & editing, Visualization. K. Keerthi: Writing review & editing. Anam Giridhar Babu: Writing review & editing. Vishaka Goyal: Writing review & editing. Lohit EA: Writing review & editing. Jyothi N. V.V.: Writing review & editing. K. Praveena: Writing review & editing. Kasibhatta Sivakumar: Writing review & editing. T.G. Satheesh Babu: Writing review & editing, Visualization. P.V. Suneesh: Writing review & editing. Hari Bandi: Writing review & editing, Visualization. Daniel-Ioan Stroe: Writing review & editing, Visualization. Sada Venkateswarlu: Conceptualization, Writing review & editing, Writing original draft, Visualization. Aristides Bakandritsos: Writing review & editing, Visualization, Supervision, Funding acquisition. Rajenahally Jagadeesh: Writing review & editing, Visualization, Validation, Supervision, Conceptualization. Radek Zboril: Writing review & editing, Validation, Supervision, Project administration, Funding acquisition, Conceptualization.

# **Data Statement**

No data were generated for the production of this work.

# **Declaration of Competing Interest**

The authors declared no conflict of interest

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### **Data Availability**

Data will be made available on request.

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Sowjanya Vallem earned her Ph.D. in Physics from the National Institute of Technology Karnataka, India, in 2019. She is currently a postdoctoral researcher at Amrita Vishwa Vidyapeetham, working on advanced electrochemical sensors for glucose and chlorine detection, aimed at healthcare and environmental monitoring. Her expertise spans nanomaterials, electrocatalysis, and energy storage, with previous research focused on Li-chalcogenide batteries and supercapacitors. Dr. Vallem's interdisciplinary work bridges materials science and applied technologies, contributing to next-generation diagnostic and sustainable energy solutions.



Sibi Malayil Gopalan earned his Ph.D. in Chemical Science from AcSIR-IIP Dehradun in 2016. He held postdoctoral positions at the University of the Western Cape, Université Laval, and Sungkyunkwan University under the BK21 + program. Currently, he is a Marie Skłodowska-Curie Actions (MSCA) fellow and senior researcher at VSB-Technical University of Ostrava, Nanotechnology Centre. Mentored by Prof. Radek Zbořil, his research focuses on the design of single-atom heterogeneous catalysts for sustainable catalytic processes, including the production of sustainable aviation fuel (SAF) from biomass and CO<sub>2</sub>.



K. Keerthi obtained her M.Sc., degree from Sri Padmavati Mahila Visvavidyalam, Tirupati, India in 2019. She is currently pursuing Ph. D under the supervision of Prof. N.V.V. Jyothi at Sri Venkateswara University. Her research work focuses on the development of efficient and eco-friendly catalysts for Oxygen evolution reaction and Hydrogen evolution reactions.



Giridhar Babu Anam is an Associate Professor in the Department of Basic Sciences at SR University, Telangana, India. He received his Ph.D. from the Department of Biotechnology and Environmental Sciences, Thapar University, India in 2011 and has held research positions at leading South Korean institutions including Chonbuk National University, Kangwon National University, and the Pohang Institute of Science and Technology. Currently, his research focuses on cyanobacterial bloom hazards, cyanophage and chemical based elimination strategies, microbial metagenomics, and renewable energy. He aims to develop sustainable solutions for pollution control and environmental remediation, effectively bridging the gap between academia, industry, and policy-making.



Vishakha Goyal completed her Ph.D. degree in catalysis and organic transformations at Academy of Scientific and Innovative Research (AcSIR), India in 2022. Subsequently, she joined in VSB-Technical University of Ostrava, as postdoctoral scientist in the group of Prof. Radek Zboril. Further, she worked as postdoctoral fellow in Leibniz Institute for Catalysis (LIKAT), Germany in 2024. In the following year, she started working as Marie Skłodowska-Curie Actions (MSCA) Postdoctoral Fellow-CZ in VSB-Technical University of Ostrava. Her research interest focuses on designing on homogeneous and heterogeneous (single atom catalysts) catalysts for sustainable organic synthesis and energy technologies.



EA Lohith received his M.Sc degree from Sri Venkateswara University, Tirupati, India in 2019. He is currently pursuing Ph. D under the supervision of Prof. N.V.V. Jyothi at Sri Venkateswara University. His research work focuses on the development of sustainable catalysts for energy sector particularly in hydrogen production through water splitting.



N.V.V. Jyothi obtained her Ph.D. in Chemistry from Sri Venkateswara University, Tirupati, in 2003. She received a UGC Postdoctoral Fellowship (UGC-PDF) in 2007 and currently serves as the Coordinator for the M.Sc. Organic Chemistry (SSC) program. In 2019, she was elected as an Associate Fellow of the Andhra Pradesh Academy of Sciences (APAS). She organized a national conference in 2017 and has published 40 research papers and two book chapters in reputed journals. Her research interests include voltammetric method development for trace metal analysis, nanomaterial synthesis, and the thermodynamic properties of binary organic liquids.



K. Praveena received her M.Sc. degree in Organic Chemistry from Sri Venkateswara University, Tirupati, India, in 2012. She worked as a lecturer at Kasturba Gandhi Balika Vidyalayas (KGBVs), Nellore, India, during 2023–2024. Currently, she is a Ph.D. scholar under the guidance of Prof. K. Sivakumar in the Department of Chemistry, Sri Venkateswara Arts College (TTD), Tirupati 517502, India. Her research interests include designing nanocatalyst for energy and environmental applications.



Kasibhatta Sivakumar is currently working as a Professor of Chemistry at S.V. Arts College (Autonomous), managed by TTD, Tirupati, India. He was awarded an M.Phil. and Ph.D. in Physical Chemistry from the same university in 1990 and 1992, respectively. His research areas include thermophysical characterization of liquid mixtures, molecular interactions in binary and ternary systems, physicochemical properties of ionic liquids, analytical determination of drugs, and biosorption of heavy metals. He has published 90 research articles and 6 book chapters in reputed national and international journals. In 2022, he was elected as a Fellow of the Royal Society of Chemistry (IJK).



T. G. Satheesh Babu, Professor at Amrita Vishwa Vidyapeetham, leads cutting-edge research in non-enzymatic biosensors, immunosensors, and POCT devices. With over 75 publications, US patents, and significant project funding, he advances sensor technologies through the Amrita Biomedical Engineering Centre, driving impactful innovations in biomedical diagnostics.



Sada Venkateswarlu received his Ph.D. in Chemistry from Sri Venkateswara University, Tirupati, India, in 2014. He worked with Prof. Minyoung Yoon (Gachon University) as a Post-doctoral Researcher and Assistant Professor (2015–2021), and with Prof. Myung Jong Kim (2021–2023). He has authored over 35 peer-reviewed publications (h-index: 21, 2000 + citations). Currently, he is a Senior Researcher in Prof. Radek Zbořil's group at the VSB–Technical University of Ostrava, Czech Republic. His research focuses on single-atom engineering, MOFs, carbon materials, and BNNTs for developing electrochemical catalysts in sensing and energy conversion applications.



Suneesh P. V., Associate Professor at Amrita Vishwa Vidyapeetham, specializes in nanomaterials for biosensors and energy storage. Active in the Biosensor Research Lab since 2008, his work focuses on developing affordable diagnostic tools, microfluidic systems, and next-generation energy devices for real-world healthcare and sustainable energy applications.



Aristeidis Bakandritsos (ORCID 0000–0003–4411–9348) is a group leader and senior researcher at CATRIN-RCPTM, Palacký University Olomouc and VSB-TUO, CEET, MELab in Ostrava, Czech Republic. He received his Ph.D. in Greece in 2006, and he was faculty member at the Dept. of Materials Science, University of Patras, before joining RCPTM. He has more than 120 peer-reviewed publications, h-index 39, and > 6000 citations (Scopus). His research is focused on the synthesis and functionalization of nanomaterials targeting advances in the fields of energy storage, catalysis, environmental remediation and biomedicine.



Hari Bandi earned his Bachelor of Science (B. Sc, 2015) and Master of Science (M. Sc, 2017) degrees in condensed matter physics from Sri Venkateshwara University, India. At present, he is pursuing a Ph.D. degree at Kyung Hee University in the Republic of Korea, where he is under the supervision of Professor Jae Su Yu. His primary research interests include transition metal oxides for the development of high-performance lithium-ion batteries, zinc-ion batteries, and supercapacitors.



Jagadeesh Rajenahally is currently a Group Leader at Leibniz Institute for Catalysis (LIKAT), Germany and Nanotechnology Center, CEET, VSB-Technical University of Ostrava, Czech Republic. He obtained PhD (2006) in Chemistry from Bangalore University, India. After having performed Postdoctoral Research (2006–2008) at the University of Florida, USA and worked as Assistant Professor (2009–2010) at VIT University, India he joined the Leibniz Institute for Catalysis in the year 2010 initially as a Scientist in the Department of Prof. Matthias Beller and then became Group Leader. His research focuses on catalysis, sustainable organic synthesis, valorization of renewable feedstocks and recycling/upscaling of wastematerials.



Daniel-I. Stroe, Associate Professor at Aalborg University, leads the Batteries Research Group, specializing in lithium-ion battery testing, predictive modeling, diagnostics, and lifetime estimation. With over 270 peer-reviewed publications, his work advances energy storage for grid and e-mobility applications. His research supports UN SDGs through sustainable, data-driven battery technologies and modeling.



Radek Zbořil is the Scientific Director of RCPTM division of Czech Advanced Technology and Research Institute (CATRIN) at Palacky University Olomouc and a head of the Materials-Envi Lab at VSB-Technical University Ostrava, Czech Republic. After receiving PhD in 2000 from Palacký University he stayed at several universities including the University of Delaware and the University of Tokyo. He was a Founding Director (2010–2020) of the RCPTM. He is an expert in nanotechnologies and author of > 700 papers with > 78,000 citations and 128H-index. He has appeared several times on the list of Highly Cited Researchers announced by Clarivate Analytics.